

Delia J. Milliron The Molecular Foundry, Lawrence Berkeley National Lab

Preparative Strategies in Solid State and Materials Chemistry UCSB-ICMR Summer School August 11, 2010





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Government warning:

I work at a National Lab and do not teach classes for a living. Therefore, I made a lot of new slides for this Summer School. Probably, there are some errors or typos. Failure to check primary sources before applying formulae could result in confusion or other mental health problems.





Lecture I: Fundamentals of nanocrystal synthesis

- Basic apparatus & techniques
- Minimizing polydispersity
- Size control
- Crystal phase control
- Lecture 2: Complex structures
 - Shape control
 - Heterostructures & chemical conversion





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Synthetic apparatus for colloidal inorganic nanocrystals





Transmission electron microscopy (TEM) of colloidal nanocrystals



20 nm

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2 nm

20 nm



Transmission electron microscopy (TEM) of colloidal nanocrystals















X-ray diffraction of MgO nanocrystals







 $d \simeq \frac{\lambda}{\beta_{corr} \cos \theta}$

Debye-Scherrer broadening



X-ray diffraction of MgO nanocrystals



Optical spectroscopy of semiconductor nanocrystals (Quantum Dots)





size













Optical spectroscopy of semiconductor nanocrystals









Workstation for Automated Nanomaterials Discovery and Analysis



Workstation for Automated Nanomaterials Discovery and Analysis



High-throughput characterization



Absorption, Emission

Size, size distribution, concentration, doping



X-ray diffraction Crystal structure, size, shape





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originally: LaMer (1950)



Classical model for colloidal nucleation and growth



originally: LaMer (1950)



Classical model for colloidal nucleation and growth



originally: LaMer (1950)











What is the critical supersaturation (S_c) ?







What is the critical supersaturation (S_c) ?





Critical radius must be small enough for stable nuclei to form

r

$$r_{c} = \frac{2\gamma V_{m}}{R(\ln S)}$$

$$\frac{dN}{dt} = A \exp\left[-\frac{\Delta G_{c}}{kT}\right] \qquad \Delta G$$

$$\frac{dN}{dt} = A \exp\left[-\frac{16\pi\gamma^{3}V_{m}^{2}}{3k^{3}T^{3}N_{A}^{2}(\ln S)^{2}}\right] \qquad 0$$

$$O(ritical radius must be small enough for stable nuclei to form$$

$$V = h + r^{2} \qquad r_{c}$$

Nucleation rate must exceed dissolution rate

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r

Diffusion controlled growth and the "focusing" effect

review by: Hyeon, et al. Angew. Chem. (2007)

Arrival rate of monomers goes as r²

Volume goes as r³

dr dt

Diffusion controlled growth and the "focusing" effect

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Diffusion controlled growth and the "focusing" effect

Arrival rate of monomers goes as r²

Volume goes as r³

Size distribution becomes narrower under diffusioncontrolled growth

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originally: Reiss (1951)

k_d monomer diffusion

D

 $\Delta \mu \propto \frac{\gamma}{r}$

originally: Ostwald (1901)

review by: Hyeon, et al. Angew. Chem. (2007)

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Wednesday, August 11, 2010

k_p

Gibbs-Thomson effect and Ostwald

Smaller particles have higher chemical potential due to surface energy

originally: Ostwald (1901)

review by: Hyeon, et al. Angew. Chem. (2007)

Gibbs-Thomson effect and Ostwald

Smaller particles have higher chemical potential due to surface energy

Dissolution of small particles, growth of large particles

originally: Ostwald (1901)

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Gibbs-Thomson effect and Ostwald

Smaller particles have higher chemical potential due to surface energy

 $\Delta \mu \propto \frac{\gamma}{r}$

Dissolution of small particles, growth of large particles

Size distribution becomes broader due to Ostwald ripening

originally: Ostwald (1901)

review by: Hyeon, et al. Angew. Chem. (2007)

Diffusion-controlled growth considering Gibbs-Thomson effect

Case of diffusioncontrolled growth with size-dependent solubility

 Curve shape is modified when considering reaction rates, but qualitatively holds

r/r_c







r/r_c





r/r_c





r/r_c











Control strategies: Achieve burst nucleation by "hot injection"











Alivisatos, et al. J. Am. Chem. Soc. (2006)





Focusing by multiple hot injections



Alivisatos, et al. JACS (1998)







Focusing by multiple hot injections



Alivisatos, et al. JACS (1998)







Focusing by multiple hot injections



Alivisatos, et al. JACS (1998) BUT... growth rates are orders of magnitudes slower than the expected diffusion limit?!?







E. Chan, et al *Nano Lett* (2010).







E. Chan, et al *Nano Lett* (2010).







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Bawendi, et al. *J. Am. Chem. Soc.* (1993) Talapin, Weller, et al. *J. Phys. Chem. B* (2001)







Bawendi, et al. *J. Am. Chem. Soc.* (1993) Talapin, Weller, et al. *J. Phys. Chem. B* (2001)

Growth accompanied by "defocusing"







Growth accompanied by "defocusing"

 Monodisperse particles isolated by size-selective precipitation

Bawendi, et al. *J. Am. Chem. Soc.* (1993) Talapin, Weller, et al. *J. Phys. Chem. B* (2001)







Growth accompanied by "defocusing"

Monodisperse
particles isolated by
size-selective
precipitation

High yield overall, but small yield of monodisperse particles.

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Bawendi, et al. *J. Am. Chem. Soc.* (1993) Talapin, Weller, et al. *J. Phys. Chem. B* (2001)



Arresting growth during focusing



Talapin, Weller, et al. J. Phys. Chem. B (2001)





Arresting growth during focusing



Talapin, Weller, et al. J. Phys. Chem. B (2001)

During focusing, size increases very rapidly

Makes size reproducibility challenging





Arresting growth during focusing



Talapin, Weller, et al. J. Phys. Chem. B (2001)

During focusing, size increases very rapidly

Makes size reproducibility challenging

Overall yield is substantially reduced because reaction is incomplete













Controlling nucleation to control size

Soon after nucleation, number of particles nearly constant in time











- Soon after nucleation, number of particles nearly constant in time
- When yield is consistent, can use number of nuclei to tune size











Number of nuclei can be controlled by supply rate of monomers

$$N \propto \frac{QV_m}{\nu} \propto \frac{QRT}{D\gamma C_e q}$$

- N total number of particles nucleated
- **Q** monomer supply rate

Sugimoto (1991, 2000).





Number of nuclei can be controlled by supply rate of monomers







E. Chan, et al *Nano Lett* (2010).



$\log[Particle] = \log k_{nuc} + a \log[Cd-OLA]_0 + b \log[TOPSe]_0$



E. Chan, et al *Nano Lett* (2010).



Number of nuclei can be controlled by reaction conditions

$\log[Particle] = \log k_{nuc} + a \log[Cd-OLA]_0 + b \log[TOPSe]_0$





Number of nuclei does predict final nanocrystal size

 $\underline{Yield \left[LimitingReagent\right]_{0}}$ $N_{CdSe/particle} =$ [Particle]



E. Chan, et al *Nano Lett* (2010).



Number of nuclei does predict final nanocrystal size

$$N_{CdSe/particle} = \frac{Yield [LimitingReagent]_{0}}{[Particle]}$$
$$d_{model} = \left(\frac{6V_{m}}{\pi} \frac{N_{CdSe/particle}}{N_{A}}\right)^{\frac{1}{3}}$$

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E. Chan, et al *Nano Lett* (2010).



Number of nuclei does predict final nanocrystal size





- Basic apparatus & techniques
 - Hot solution with surfactant, precursors
 - TEM complemented by optical spectroscopy and XRD
- Minimizing polydispersity
 - Separation of nucleation and growth
 - Focusing size distribution by diffusion-controlled growth, high supersaturation
- Size control
 - Post-synthetic separation or arrested growth possible, not ideal
 - Controlling number of particles nucleated most effective




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Single component phase transitions

Phase diagram indicates equilibrium (lowest free energy) phase at each temperature and pressure

 Lines between phases represent first order phase transitions





Each material may exist in multiple crystal phases

liquid In general, multiple solid 1800 SiO₂ phases exist with cristobalite different crystal high-quartz 1500 structures tridymite **()** 200 Temperature (and **Femperature** pressure) are two 900 parameters which coesite influence the crystal 600 phase of nanocrystals low-quartz 300 stishovite 30 50 80 90 60 70100 **Pressure (kbar)**



CdTe: Wurtzite vs. Zinc Blende

wurtzite:

Both are tetrahedrallybonded, 4-coordinate

Differ only in stacking (hcp vs. fcc)

Wurtzite stable at higher temperatures

zinc blende:



Controlling phase with temperature





μÌ)

nnnn



Bulk and surface contributions to free energy – Implications for phase control

Surface energy (γ) varies
with crystal phase

$$\Delta G = 4\pi r^2 \gamma - \frac{4}{3}\pi r^3 \frac{RT\ln S}{V_m}$$

Surface Energy

Bulk Energy Surface energy increasingly important at small sizes

 Low γ phases become more stable at small sizes







Surface area (10⁴ m²/mol)



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Phase stability of TiO₂ versus size



Surface area (10⁴ m²/mol)

Navrotsky, et al. J. Chem. Thermo. (2007).

Surface enthalpy measured by calorimetry





Phase stability of TiO₂ versus size



Surface enthalpy measured by calorimetry

Stable phase depends on size

rutile in bulk, then brookite, and anatase at small sizes

Surface area (10⁴ m²/mol)



Navrotsky, et al. J. Chem. Thermo. (2007).



Phase stability of TiO₂ versus size



Surface enthalpy measured by calorimetry

Stable phase depends on size

rutile in bulk, then brookite, and anatase at small sizes

 Nanocrystals are commonly anatase, some brookite



Navrotsky, et al. J. Chem. Thermo. (2007).



Size-dependent phase transition: crystallization of amorphous GeTe







Size-dependent phase transition: crystallization of amorphous GeTe



Synthesize size-selected amorphous GeTe nanoparticles

Use XRD while heating to observe crystallization





Size-dependent phase transition: crystallization of amorphous GeTe



Synthesize size-selected amorphous GeTe nanoparticles

Use XRD while heating to observe crystallization

 Size-dependence implies amorphous state becomes relatively more stable at small sizes





Ligands can modify surface energy and thereby relative phase stability

$$\Delta G = 4\pi r^2 \gamma - \frac{4}{3}\pi r^3 \frac{RT\ln S}{V_m}$$

SurfaceBulkEnergyEnergy







Ligands can modify surface energy and thereby relative phase stability

$$\Delta G = 4\pi r^2 \gamma - \frac{4}{3}\pi r^3 \frac{RT\ln S}{V_m}$$

SurfaceBulkEnergyEnergy

$$\Delta G_{surf} = (\gamma_a A_a + \gamma_b A_B)$$







Ligands can modify surface energy and thereby relative phase stability

$$\Delta G = 4\pi r^2 \gamma - \frac{4}{3}\pi r^3 \frac{RT\ln S}{V_m}$$

SurfaceBulkEnergyEnergy

$$\Delta G_{surf} = (\gamma_a A_a + \gamma_b A_B)$$



Reduction in surface energy by ligands is specific to structure of each facet

 Different phases present different facets and can be selectively stabilized by a ligand



The (more) complete CdTe story



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Peng, et al Chem. Mater. (2003).

The (more) complete CdTe story



Peng, et al Chem. Mater. (2003).



The (more) complete CdTe story



Peng, et al Chem. Mater. (2003).







Crystal structures from Mai et al. *J. Am. Chem. Soc.*, 2006, 128 (19), 6426-6436

α-NaYF₄: Yb, Er cubic (fluorite) weak upconversion β-NaYF₄: Yb, Er hexagonal (gagarinite) strong upconversion





X-ray diffraction







X-ray diffraction





Oleic acid preferentially stabilizes the cubic (α) phase



* NaF byproduct



X-ray diffraction



β-NaYF₄ fraction from Rietveld refinement



Oleic acid preferentially stabilizes the cubic (α) phase



* NaF byproduct



Temperature

Bulk phase diagrams provide guidance, but exclude surface energy contributions

Size

Phases with lower surface energy (γ) favored at smaller sizes

Ligands

Reduce surface energy of specific facets, can stabilize specific phases

