



*Some applications of photoluminescence for  
probing polymer relaxation processes*

**Part 2**

p1

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**Slide 1**

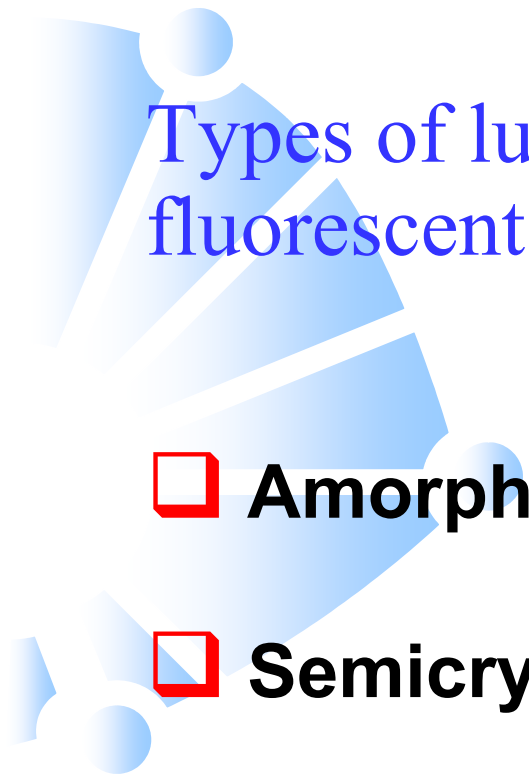
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prpg, 7/2/2005

# outline

- Polymer systems and morphology**
- Polymer motions, phase transitions and polymer relaxation processes**
- Photoluminescence spectroscopy and polymer relaxation processes**
  - fluorescence
  - phosphorescence
  - quenching processes
  - time resolved spectroscopy
- Luminescence in polymers**
  - Non-fluorescent polymers (host-guest systems)**
  - Polymers modified with luminescent groups**
  - Intrinsically luminescent polymers**
  - Conjugated luminescent polymers**
  - Electroluminescence and photoluminescence**



Types of luminescent polymers : 1. non-fluorescent polymers (host-guest systems)

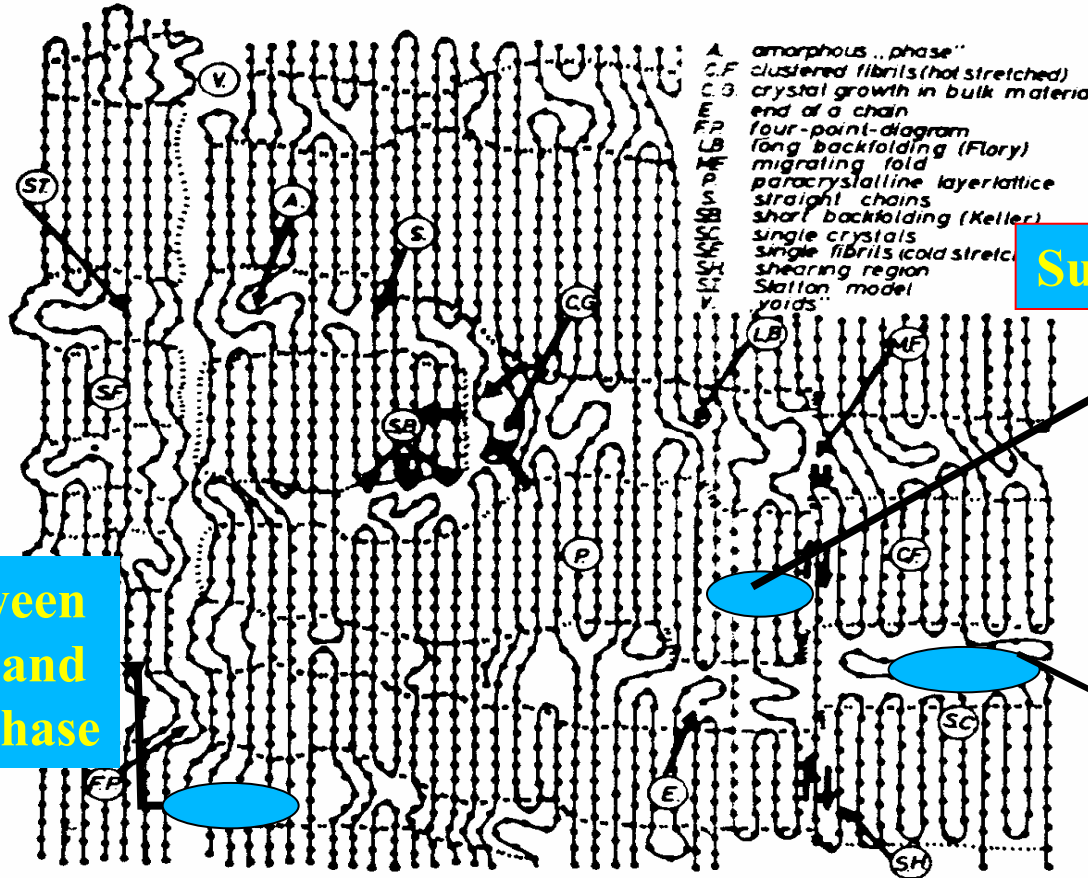
**Amorphous materials**

**Semicrystalline polymers**

How the guest interacts with the polymer?  
Which site is the guest sensing?

# Guest in semicrystalline polymers

The Reentry Problem in Lamellae



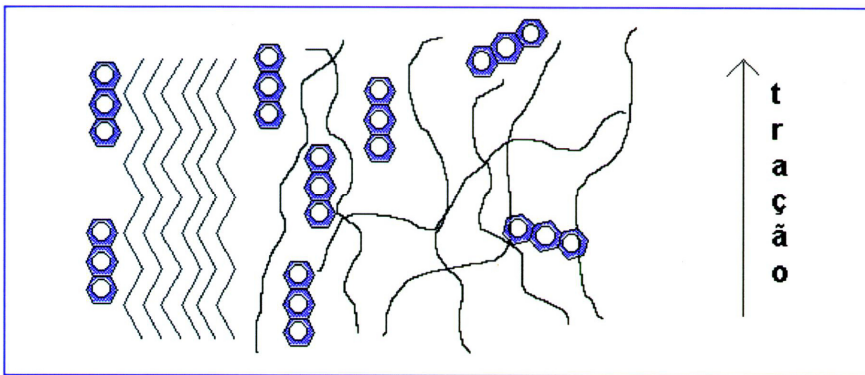
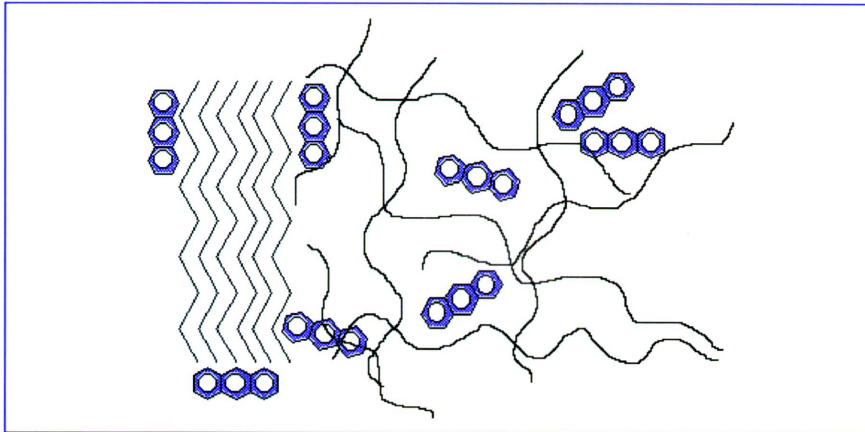
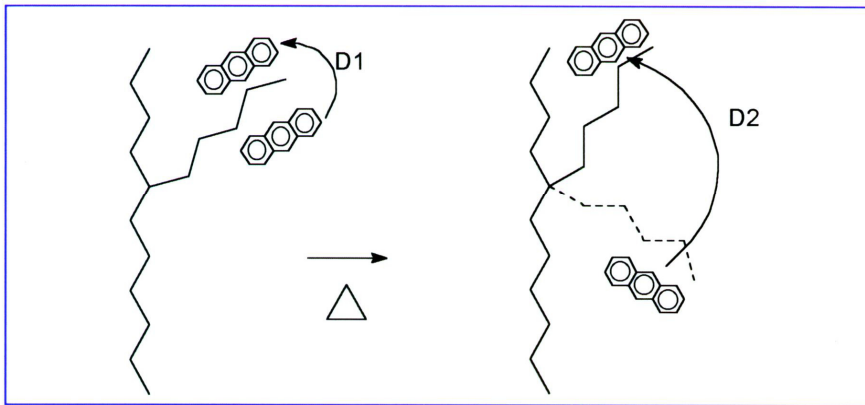
Surface of a lamellae

Interfase between amorfous and crystalline phase

Amorphous region

MODEL OF LINEAR POLYETHYLENE

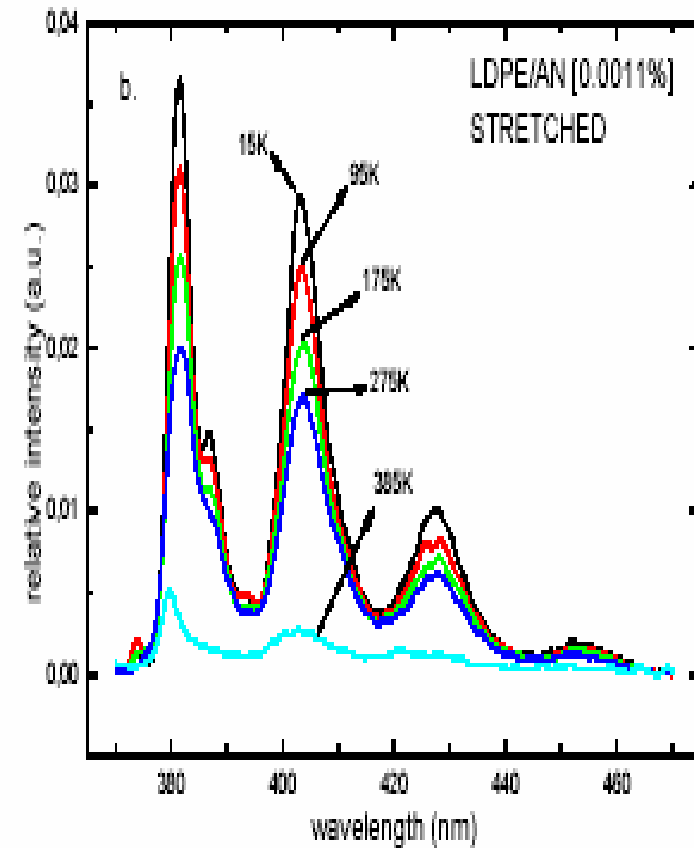
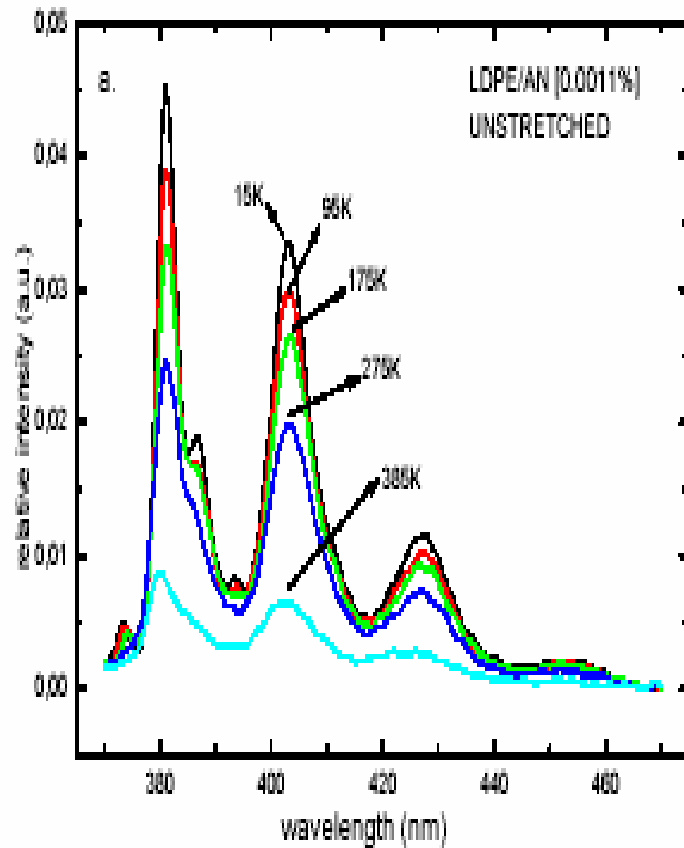
# Probing orientation in stretched materials

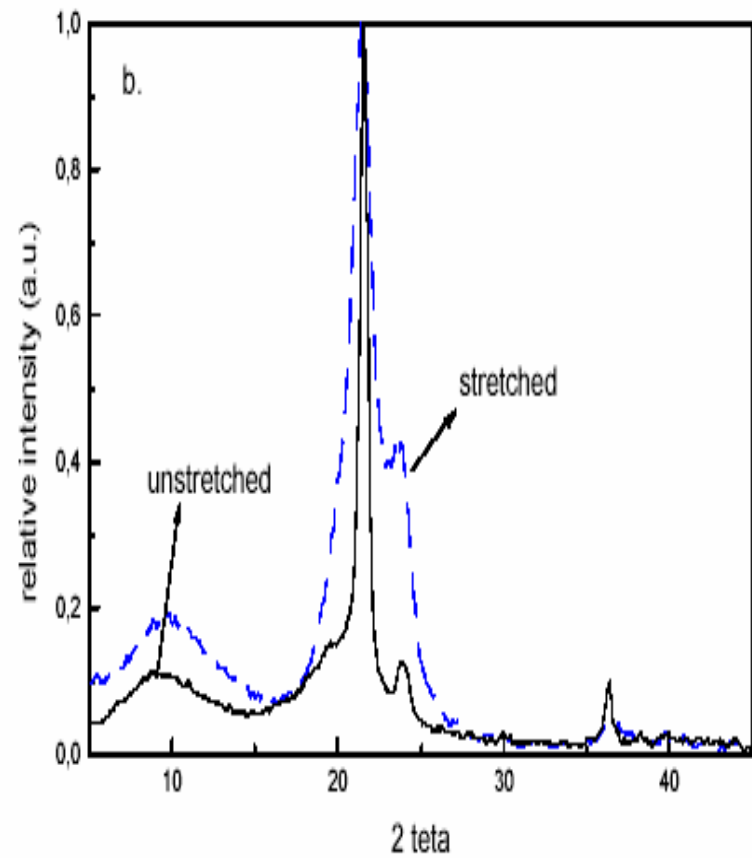
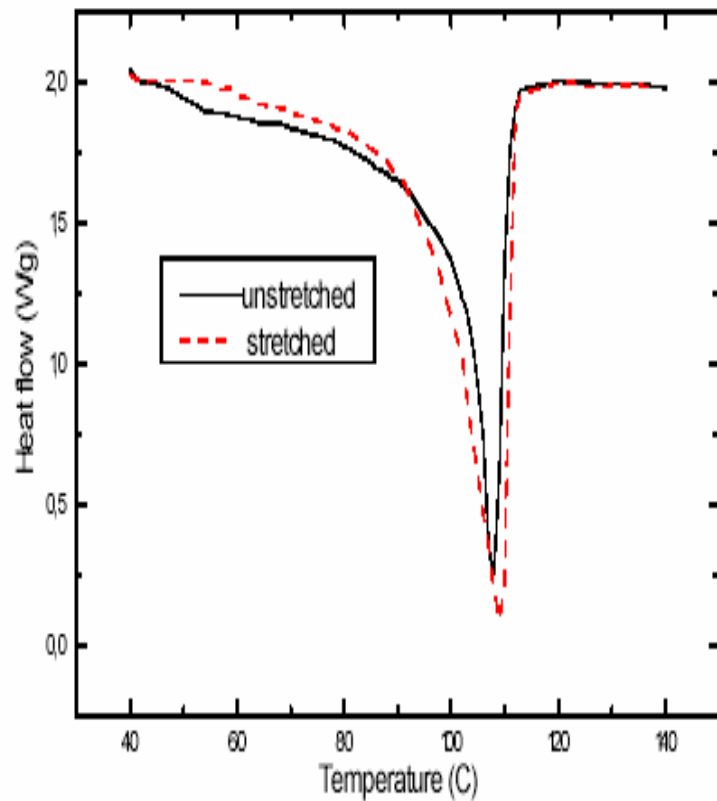


Molecules changes from one site to other  
and can be preferentially oriented in  
the stretching direction

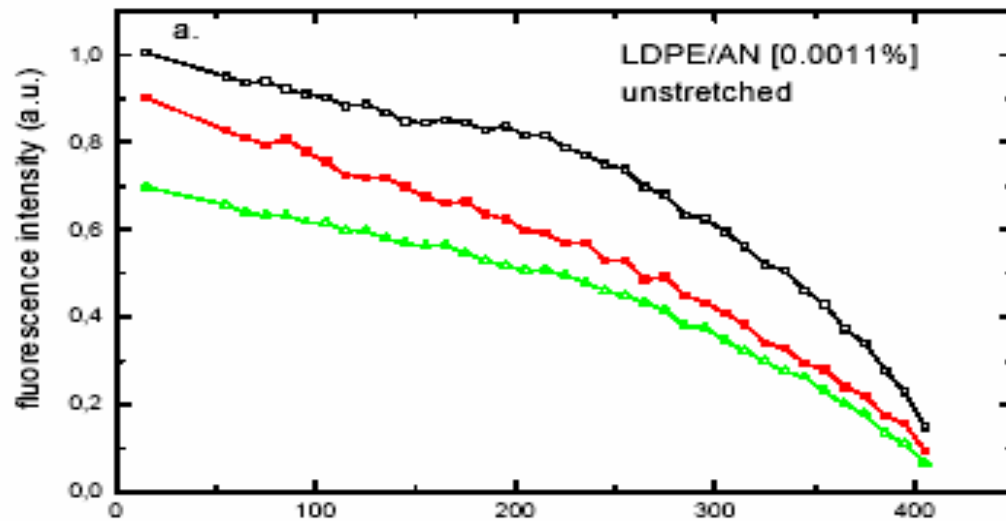
They can sense the new environment

# Relaxation processes in oriented polymers

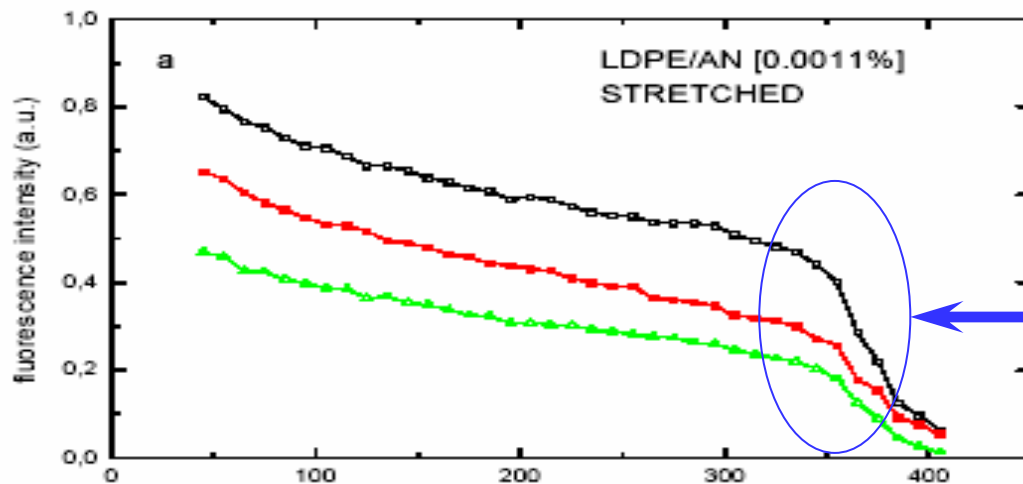








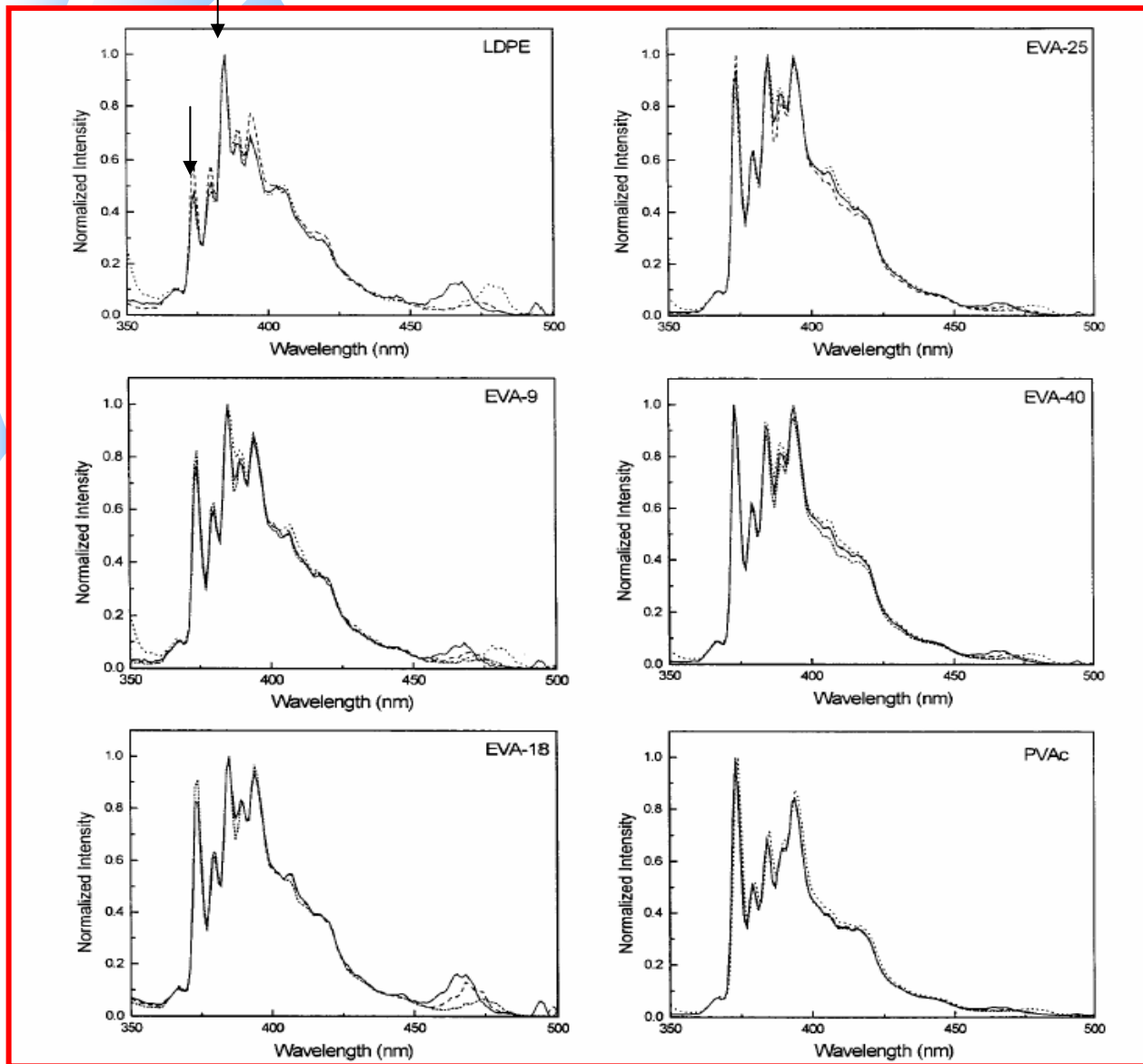
Relaxation processes associated with the interface crystalline amorphous interface are more defined



There is a larger population of anthracene molecules located in the interface compared with non-stretched material

$\alpha$ -relaxation process

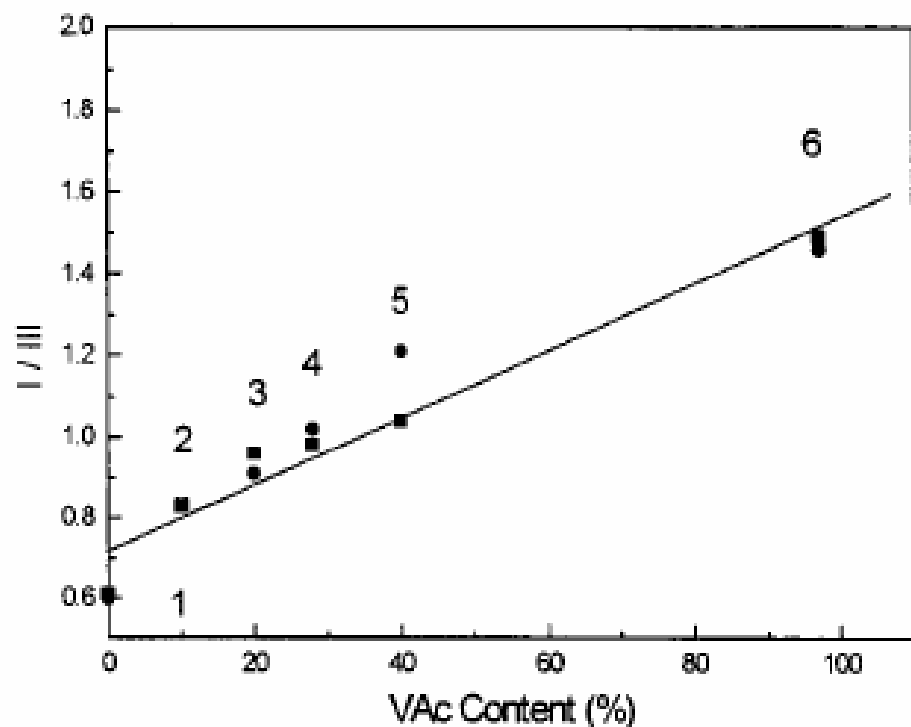
# How important is the distance for a guest sensor?



Pyrene in LDPE  
and some copolymers  
with vinyl acetate  
(EVA)

$I_I/I_{III}$  ratio changes

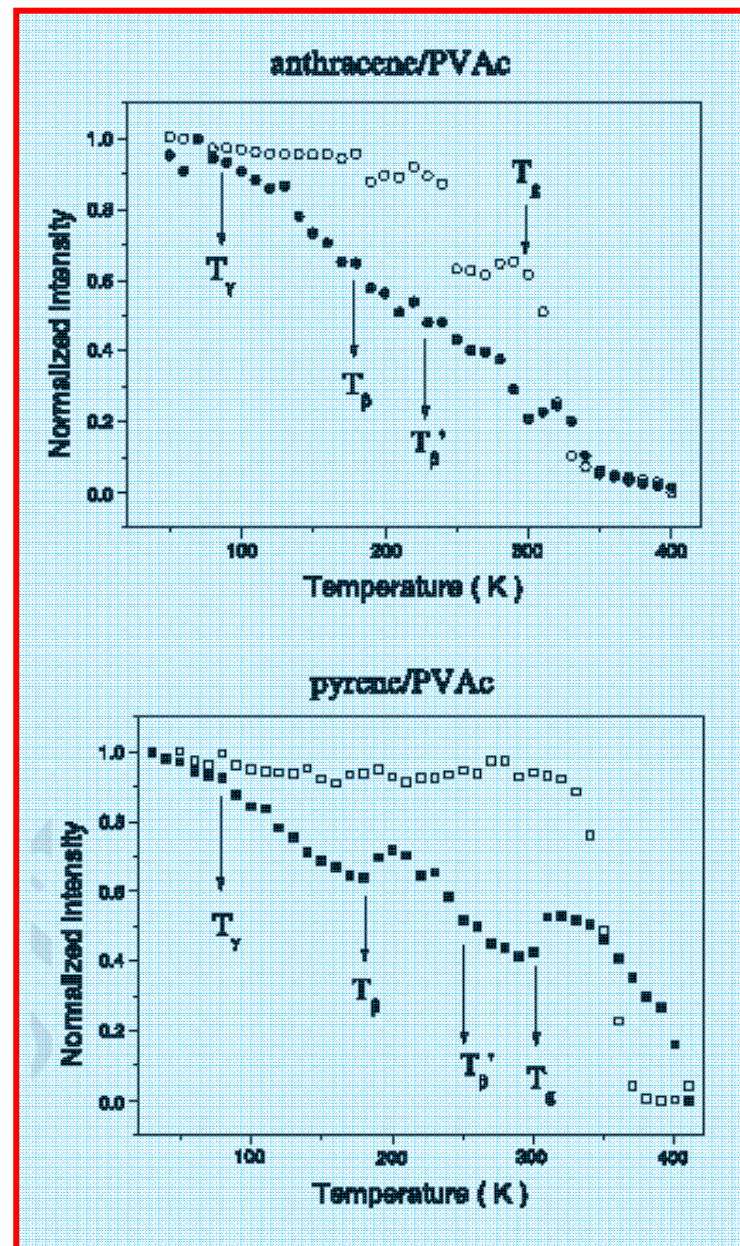
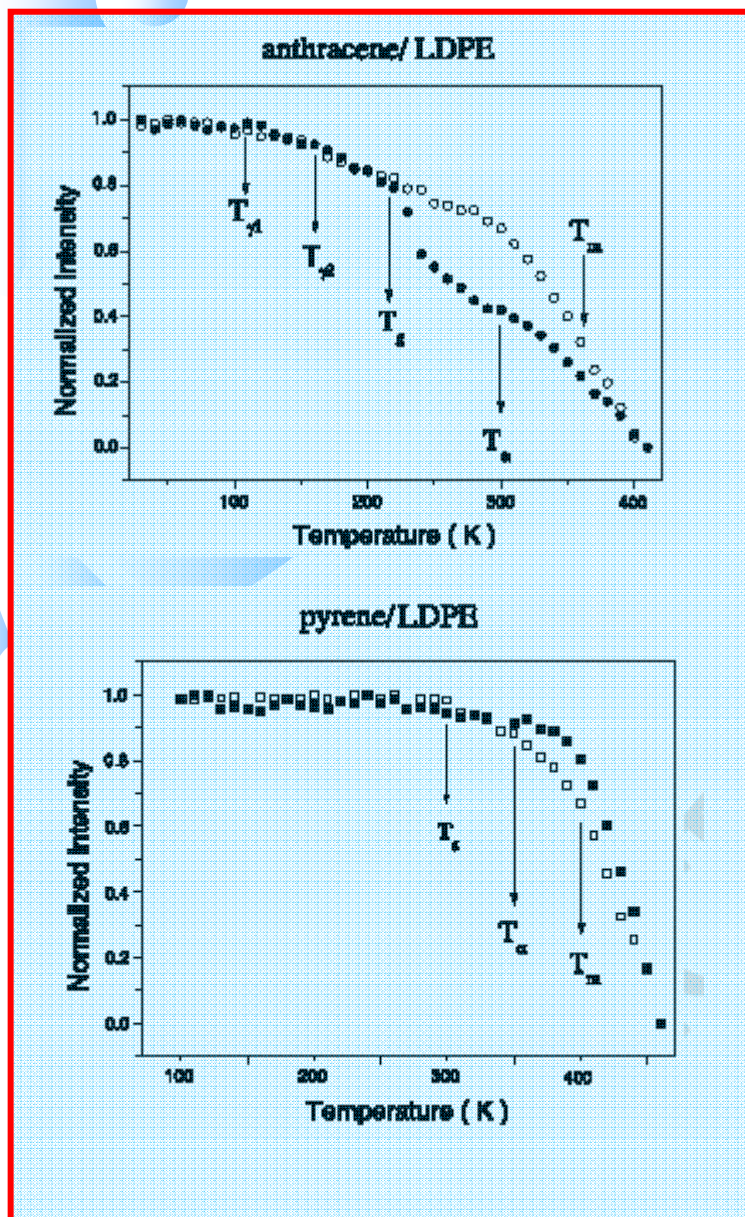
Lifetimes also change



Linear correlation  
with lifetime and  $I_I/I_{III}$  ratio  
because the random distribution  
of the polar groups

**TABLE 2: Decay Constants  $\tau_F$  (ns) and Vibrational Intensity Ratios  $I_I/I_{III}$  for Fluorescence of Pyrene Sorbed on the Surface (S) and in the Bulk (B) of Polymer Films.  $\lambda_{Em} = 393$  nm,  $\lambda_{Ex} = 336$  nm**

	LDPE		EVA-9		EVA-18		EVA-25		EVA-40		PVAc	
	S	B	S	B	S	B	S	B	S	B	S	B
$I_I/I_{III}$	0.61	0.60	0.83	0.83	0.96	0.91	0.98	1.02	1.04	1.21	1.49	1.46
$\tau_F$ (ns)	$395 \pm 1$	$392 \pm 3$	$373 \pm 3$	$370 \pm 2$	$359 \pm 1$	$358 \pm 2$	$346 \pm 2$	$347 \pm 2$	$352 \pm 1$	$348 \pm 1$	$311 \pm 3$	$299 \pm 1$

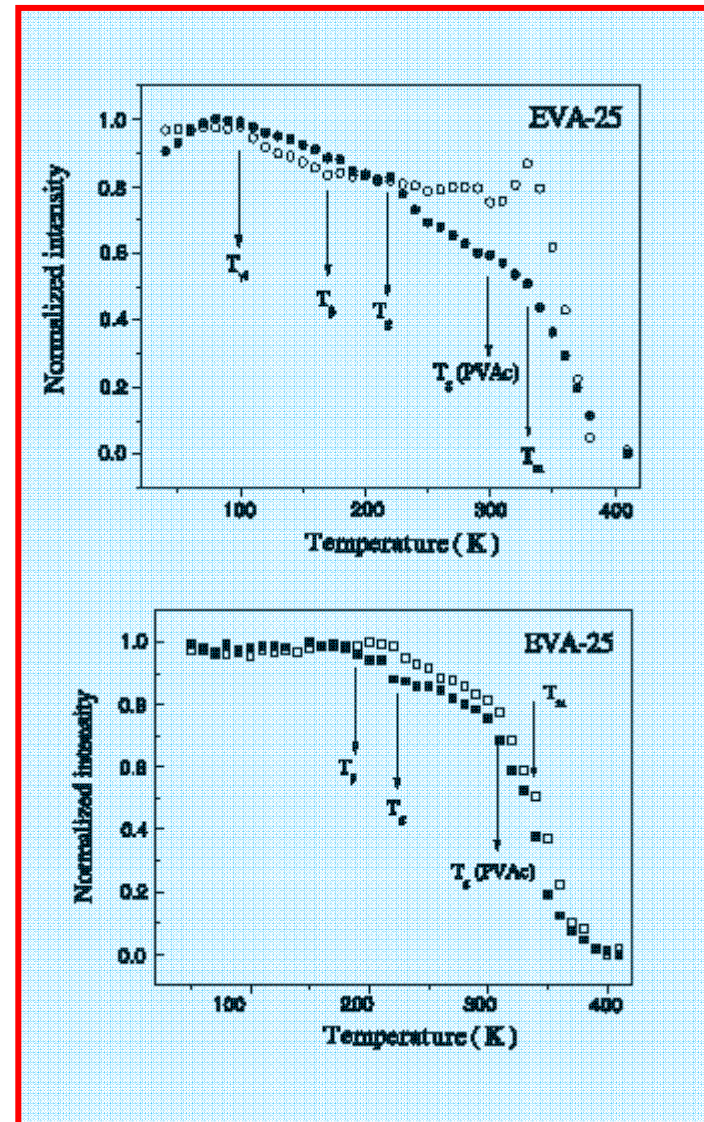
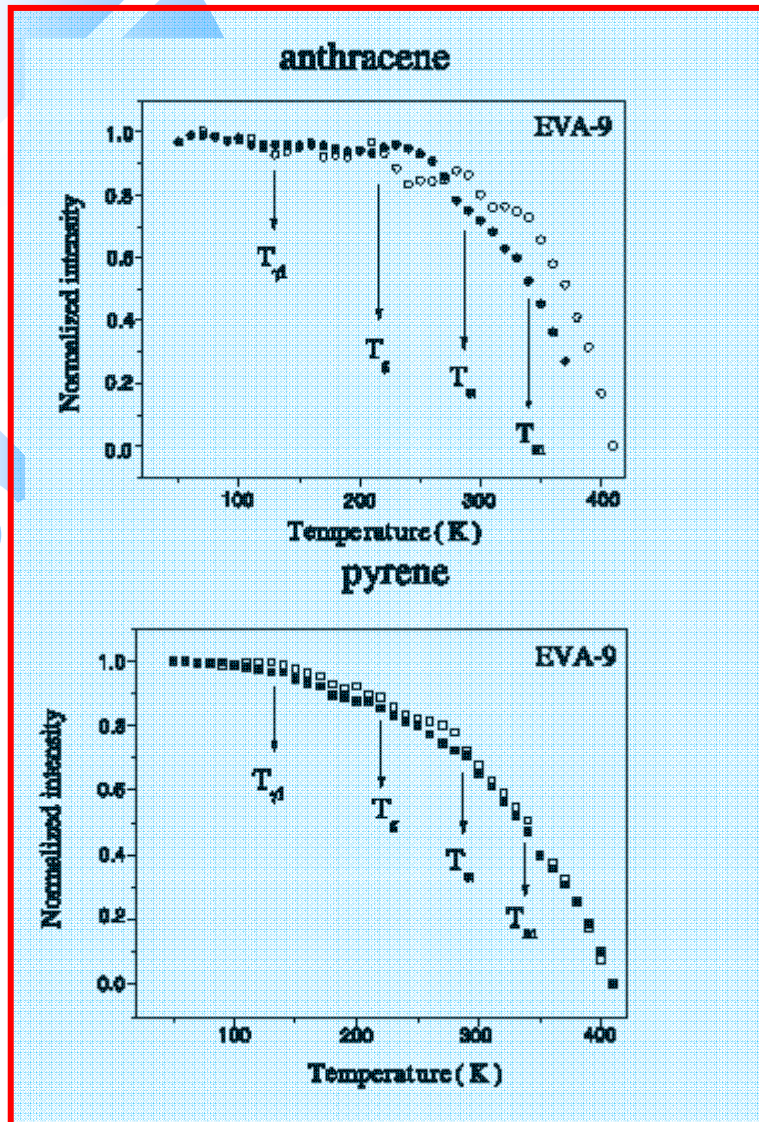


1<sup>st</sup> (open) and 2<sup>nd</sup> run (dark)

Yamaki, Eur. Polym. J., 2002

Transition and relaxation temperatures (K) (from the data shown in Figs. 4 and 5), for pyrene and anthracene sorbed in LDPE and PVAc

polymer	$T$ (K)	Assignments
LDPE	$T_{\gamma 1} = 110$	Motions of methylenic segments in the amorphous phase [18,19,21,32]
	$T_{\gamma 2} = 160$	Motions of folded chains in the interfacial regions [32]
	$T_g = 220-250$	Glass transition [18-21,32-39]
	$T_\alpha = 300$	Motions of the segments on the surface of the crystals [18-21,32-39]
	$T_m = 360$	Melting point
PVAc	$T_\gamma = 80-90$	Motions of end groups
	$T_\beta = 180-190$	Small amplitude rotary motions of one or two monomeric units in an amorphous region [40-46]
	$T_\beta = 240-250$	Rotation of the ester group leading to a <i>cis-trans</i> isomerization [9]
	$T_g = 300-310$	Glass transition [40]

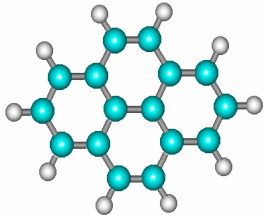


First (dark) and second (open) scans

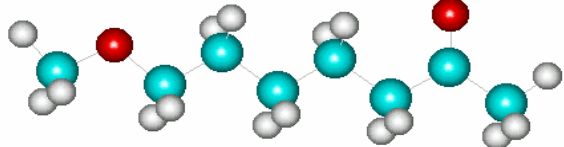
Transition temperatures (K) for EVA from the data shown in Fig. 5

Polymer	$T$ (K)	Assignment
EVA-9 and EVA-18	$T_{\gamma 1} = 110-130$	Motions of methylenic segments in the amorphous phase [18,19,21,32,36]
	$T_g = 220$	Glass transition of the LDPE segments [18-21,32-39]
	$T_\alpha = 270-290$	Motions of the chains in the interfacial region of the polymer matrix [18-21,40]
	$T_m = 330-340$	LDPE melting point
EVA-25	$T_\gamma = 100$	Motions of methylenic segments in the amorphous phase [18,19,21,32,36] and motions of end groups
	$T_\beta = 180-190$	Small amplitude rotary motions of one or two monomeric VAc units in an amorphous region [9,40-46]
	$T_g = 220$	Glass transition [18-21,32-39]
	$T_\alpha$ (PE) or $T_g$ (VAc) = 300-310	Motions of the chains in the interfacial region of the polymer matrix [18-21,40]
	$T_m = 330-340$	LDPE melting point
EVA-33 and EVA-40	$T_\gamma = 100$	Motions of end groups
	$T_\beta = 180-190$	Small amplitude rotary motions of one or two monomeric units in an amorphous region [9,40-46]
	$T_g = 230$ K	Glass transition [18-21,32-39].
	$T_g$ (PVAc) = 300-310	Mobility of the chains in the interfacial region of the polymer matrix [18-21,40] coupled with the PVAc glass transition [40]
	$T_m = 330-340$	LDPE melting point

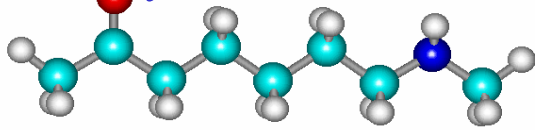
# Naylons = condensation of an amide a carboxylic acid



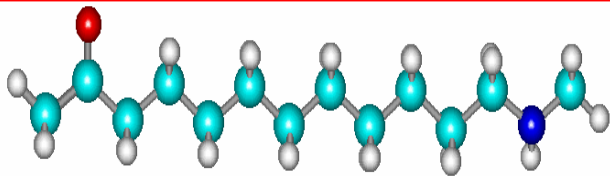
Poly(caprolactone)



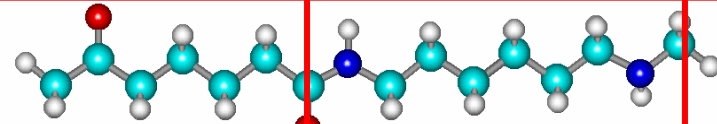
Naylon-6



Naylon-11



Naylon-6,6

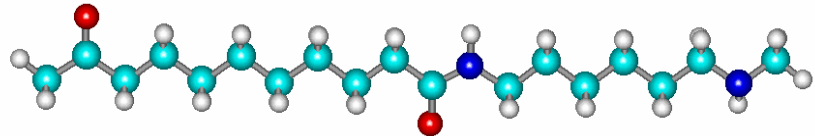


Naylon-6,9

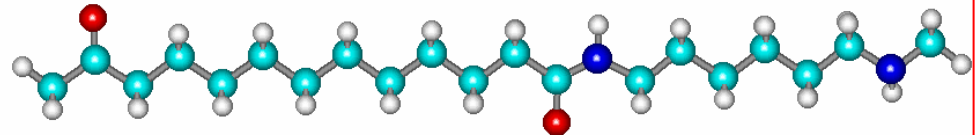


constant

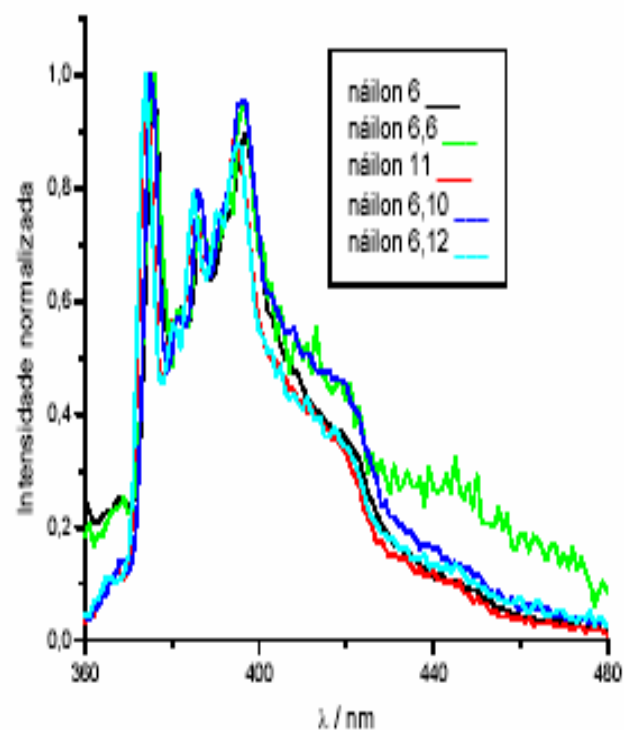
Naylon-6,10



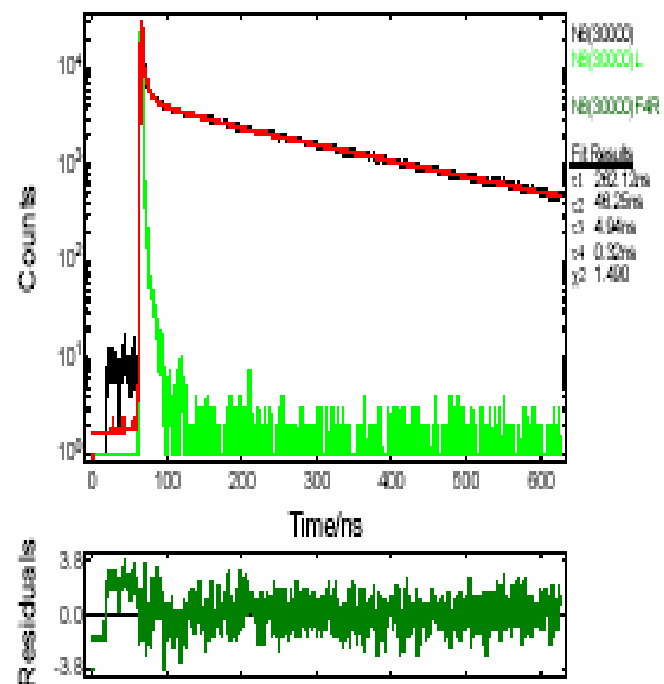
Naylon-6,12







**Figura 1.** Espectros de fluorescência fotoestacionária do pireno em diferentes náilons. A intensidade foi normalizada em relação à banda III (385 nm).



**Figura 2.** Curva de decaimento de fluorescência do pireno sorvido em matriz de náilon-6 (superior) e curva de distribuição de resíduos do cálculo de deconvolução em relação ao decaimento da lâmpada.

# Photophysical properties of pyrene in nylons

Poly(caprolactam)	1.15	265 ns
Naylon-6	1.41	260 ns
Naylon-11	1.27	276 ns
Naylon-6,6	1.44	250 ns
Naylon-6,9	1.31	276 ns
Naylon-6,10	1.30	296 ns
Naylon-6,12	1.29	310 ns

Decrease of the polarity

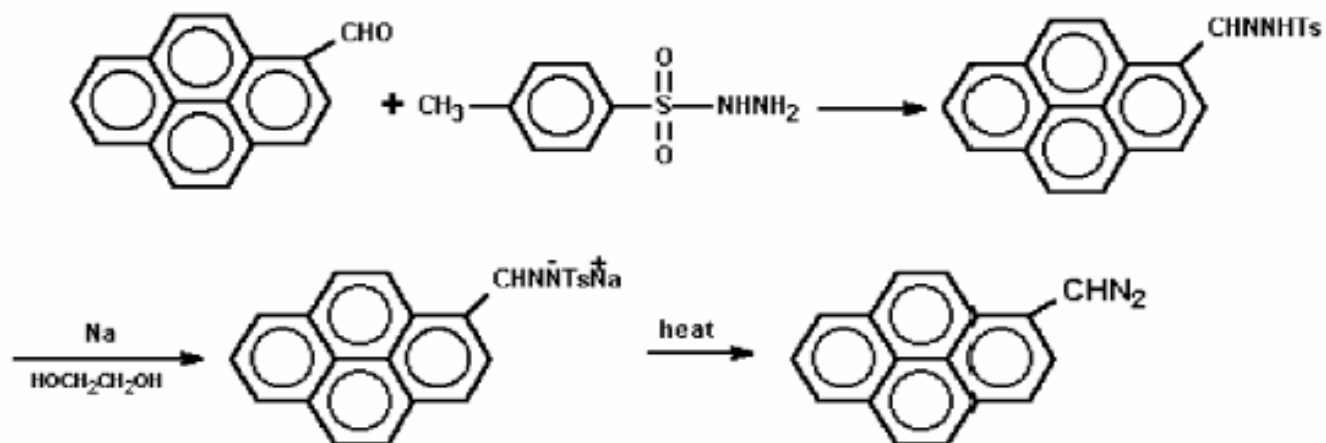




## Polymers modified with luminescent groups

- Will sense the environment around the molecule
- More sensitivity to the motions involving the segments where they were bonded
- Selective attachment should enhance the sensitivity and coupled motions can be analyzed.

# Polyethylene and vinyl acetate copolymers modified with pyrenyl groups



Scheme 1 Synthesis of PDAM.<sup>24</sup>

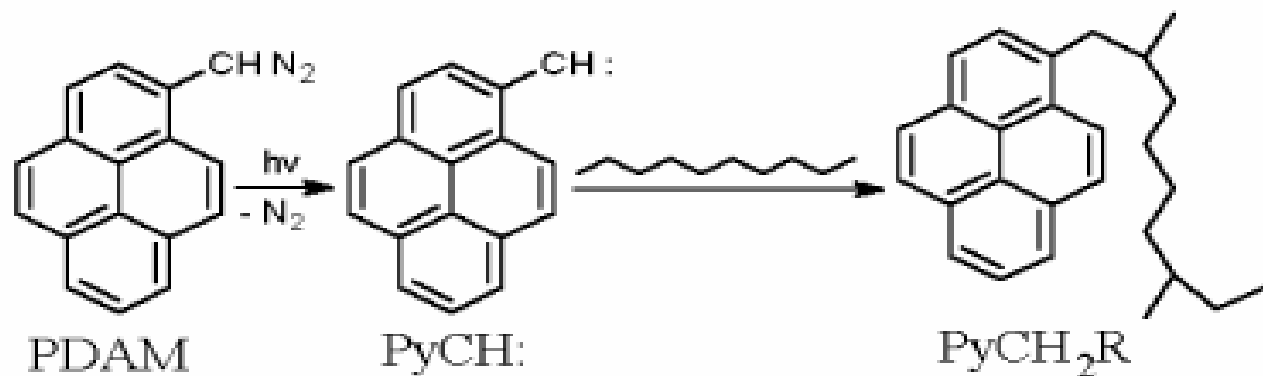
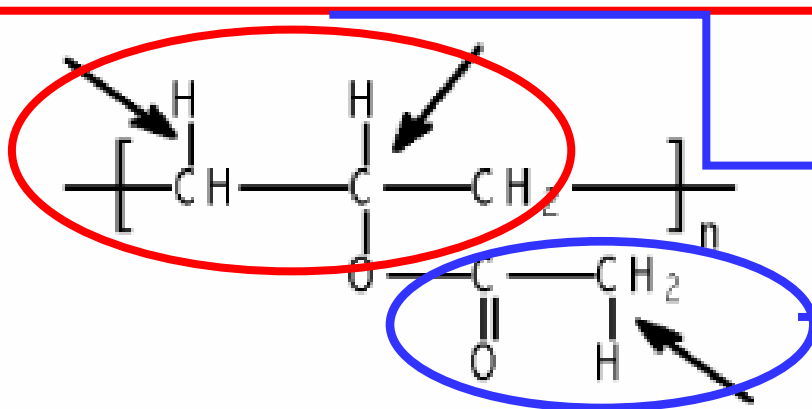


Fig. 1 Carbene insertion into 2° C-H bonds along methylene chains of polyethylene.<sup>33</sup> See text and Fig. 2 for other possibilities.

## Selective attachment

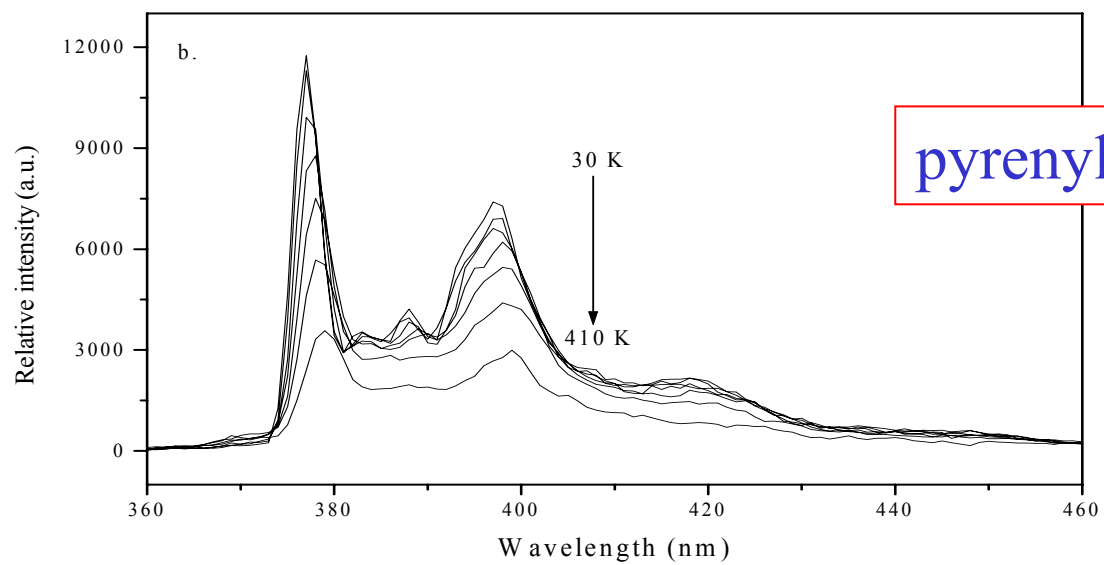
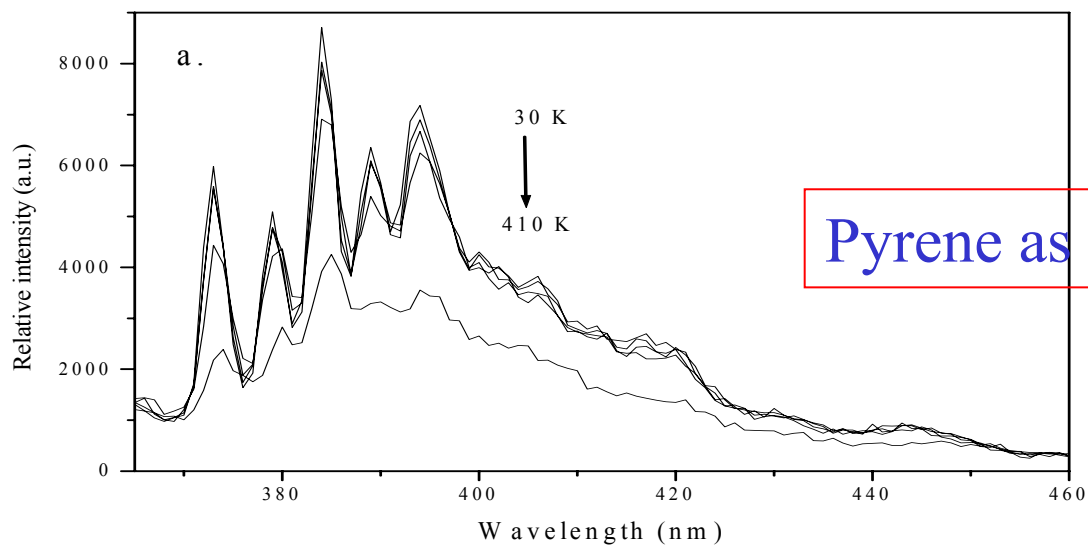
Our results demonstrate that, regardless of the complexity of the micromorphology of EVA copolymers, selective covalent attachment of fluorescent reporters to the polymeric chains can be achieved *via* reactions with pyren-1-ylcarbene, produced photochemically *in situ*.

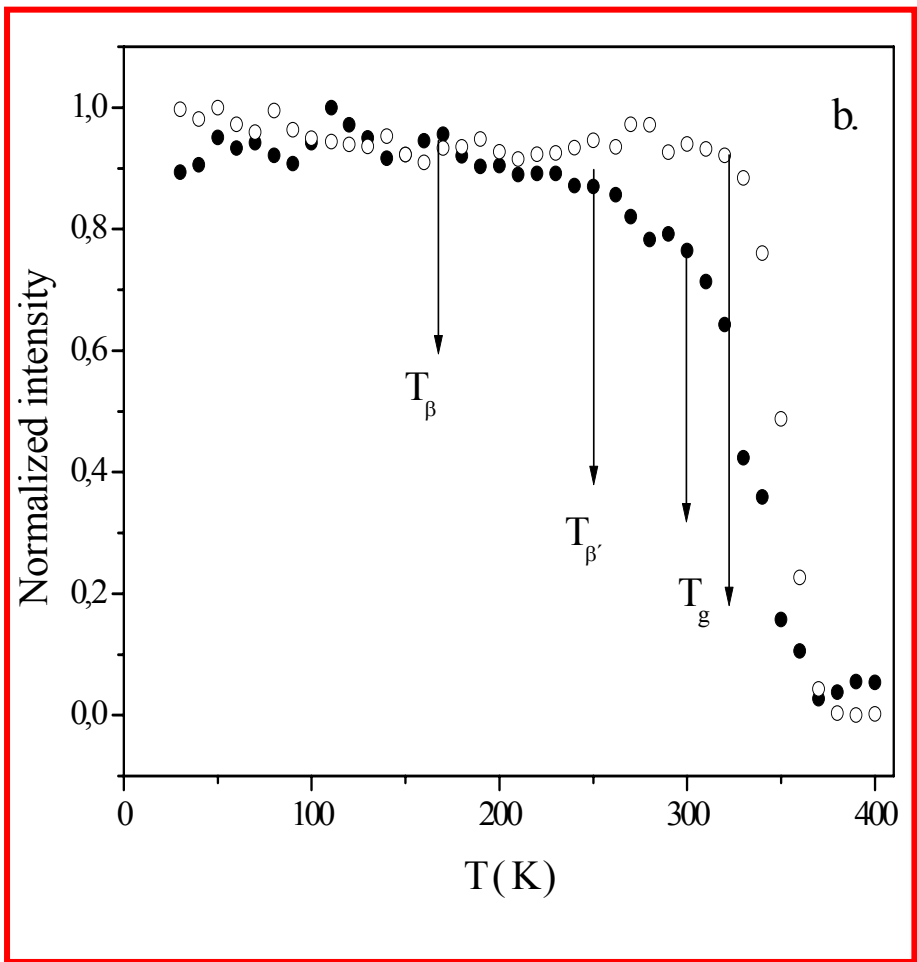
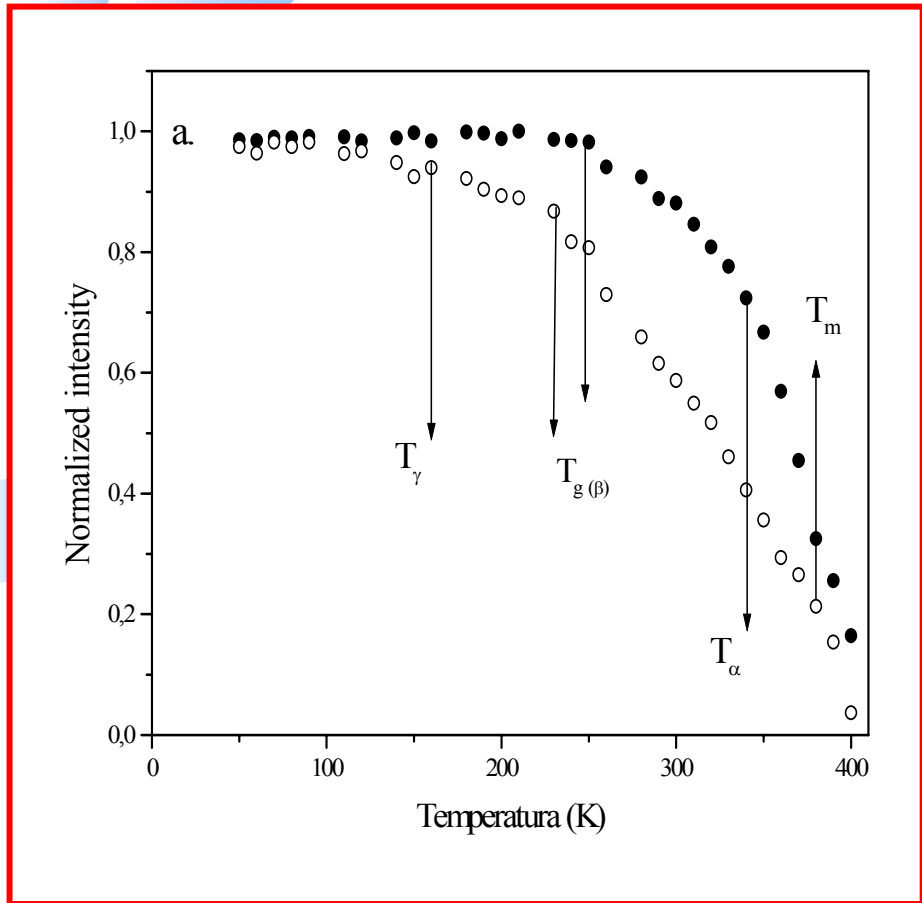


**Fig. 2** Possible attachment sites for insertion of pyrenyl groups (indicated by arrows) on pendant acetate groups of EVA and PVAc polymer chains.

Relaxation processes will be associated with these groups

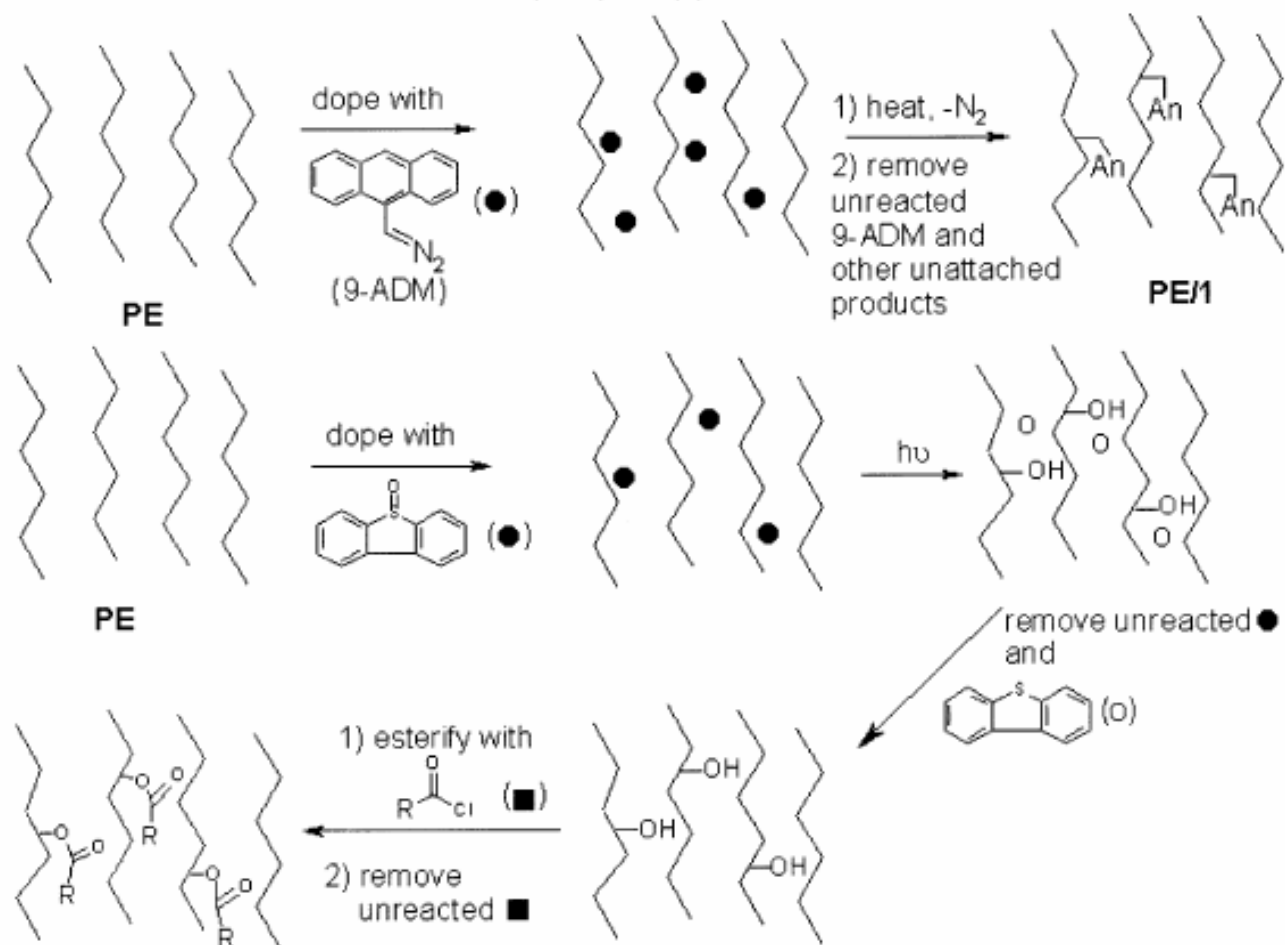
Not observed





**Normalized and integrated fluorescence intensities versus temperature for pyrene (●) and 1-pyrenyl groups (○) in (a) LDPE and (b) PVAC.**

**Scheme 1. Protocols for Covalent Attachment of 9-Anthryl (An) Groups to PE Chains; R = AnCH<sub>2</sub>- (2) and An(CH<sub>2</sub>)<sub>10</sub>- (3)<sup>11,13,24,27</sup>**





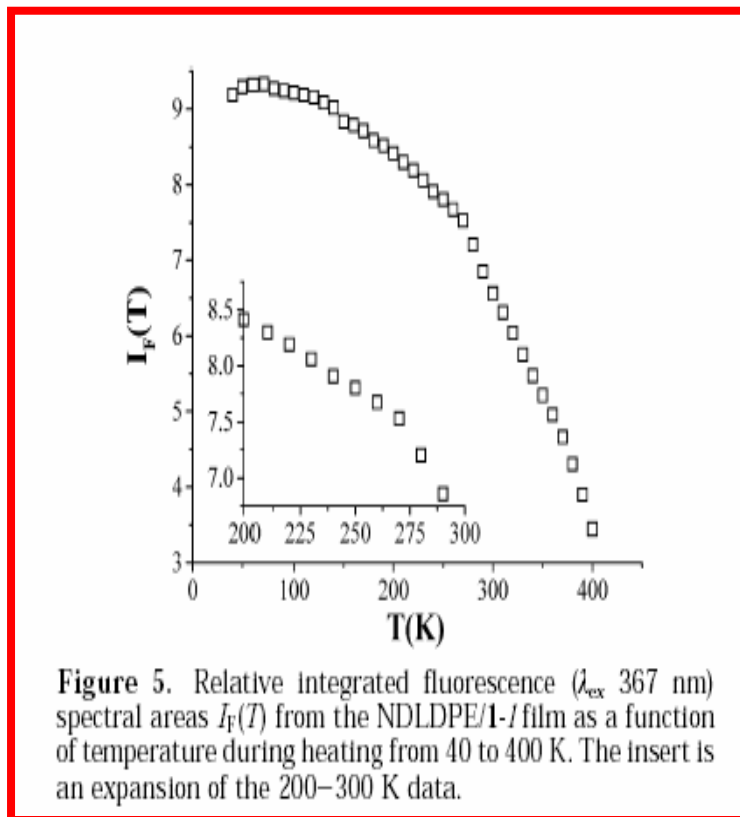
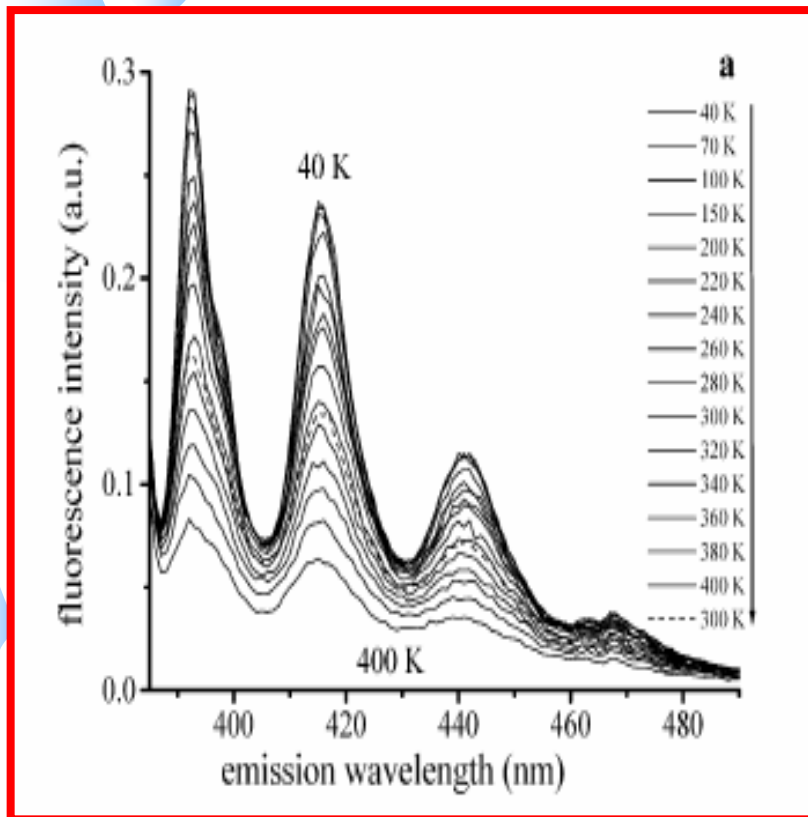
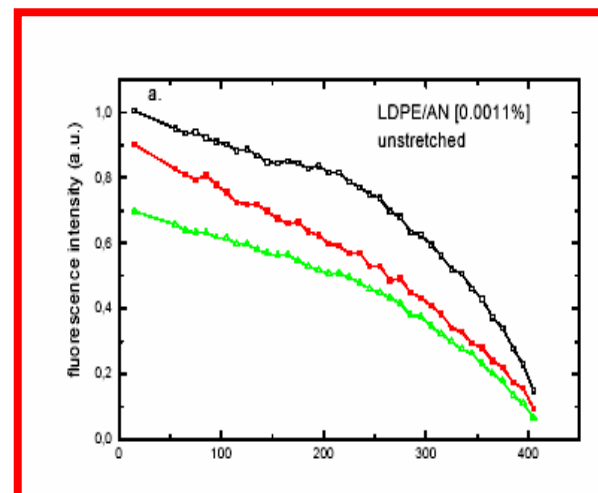
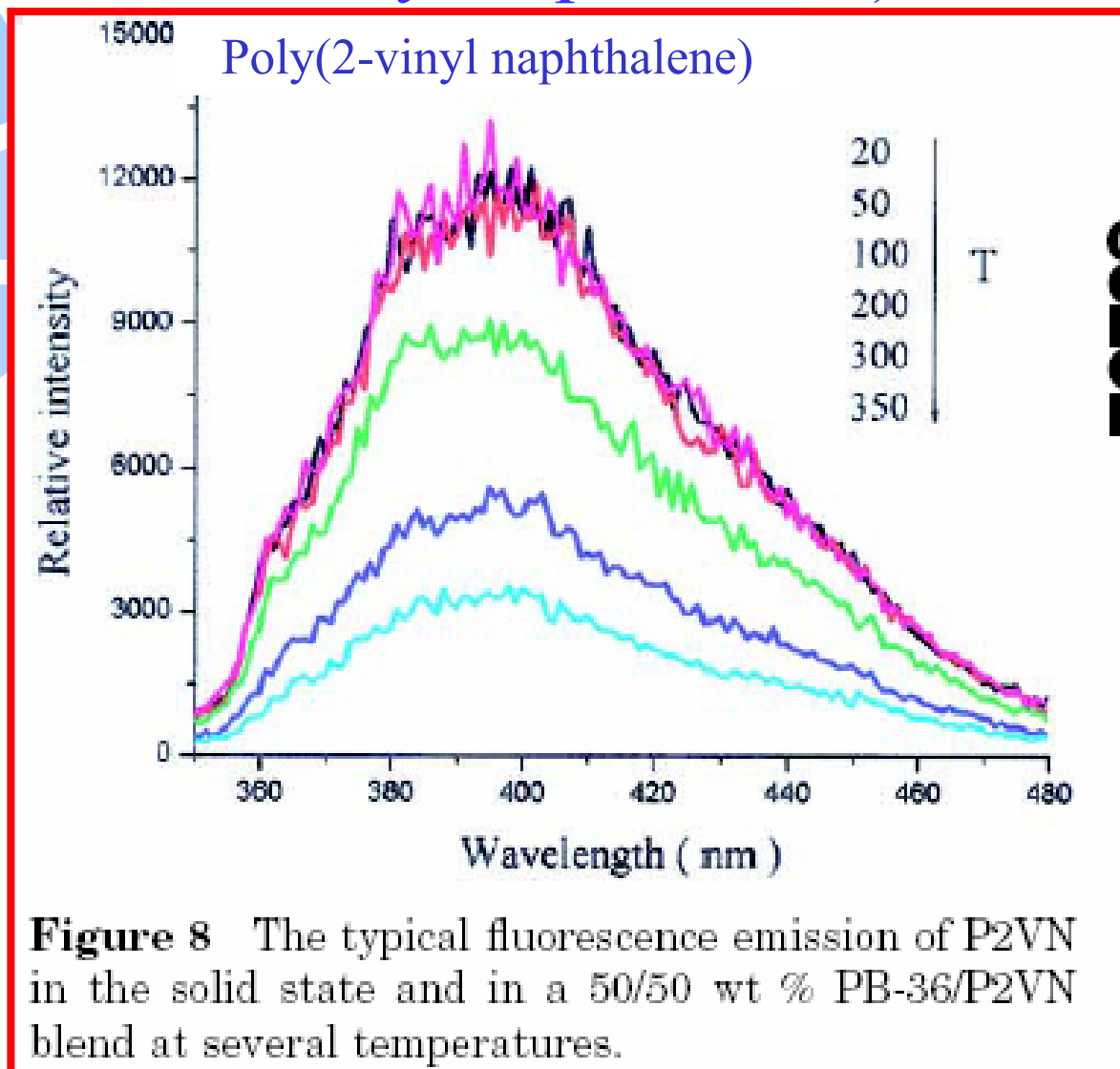
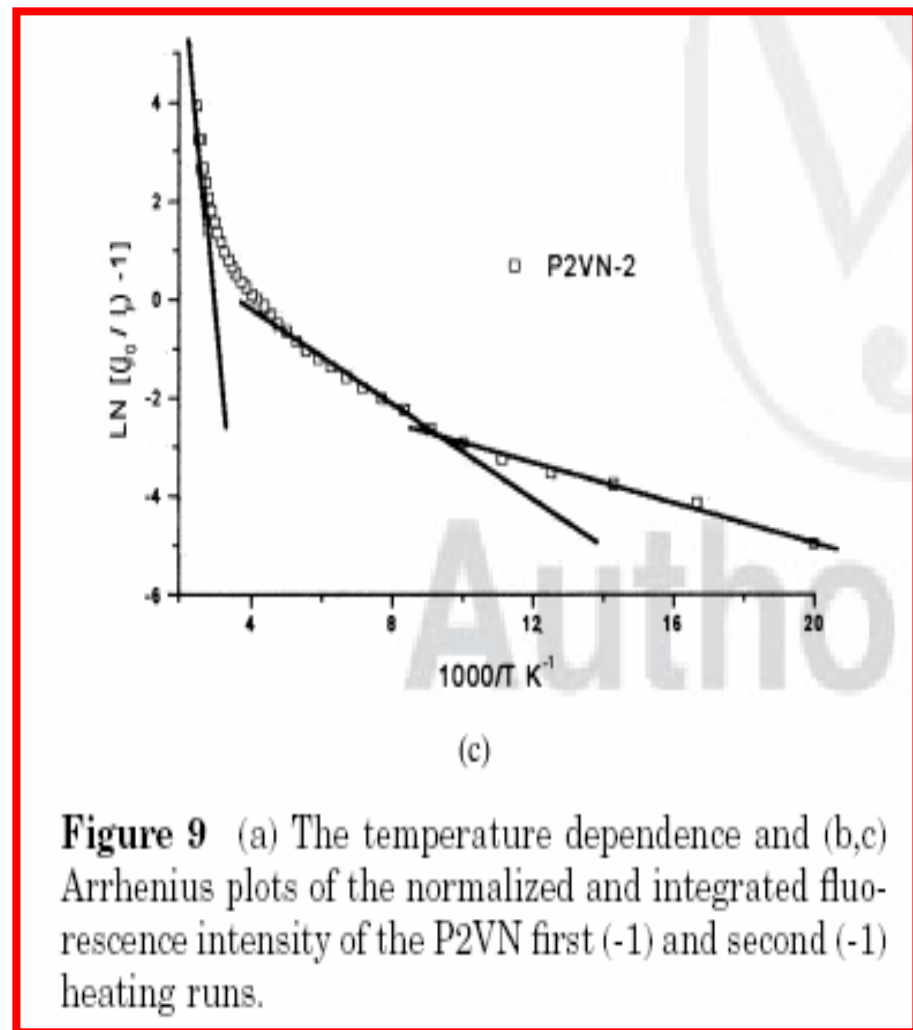
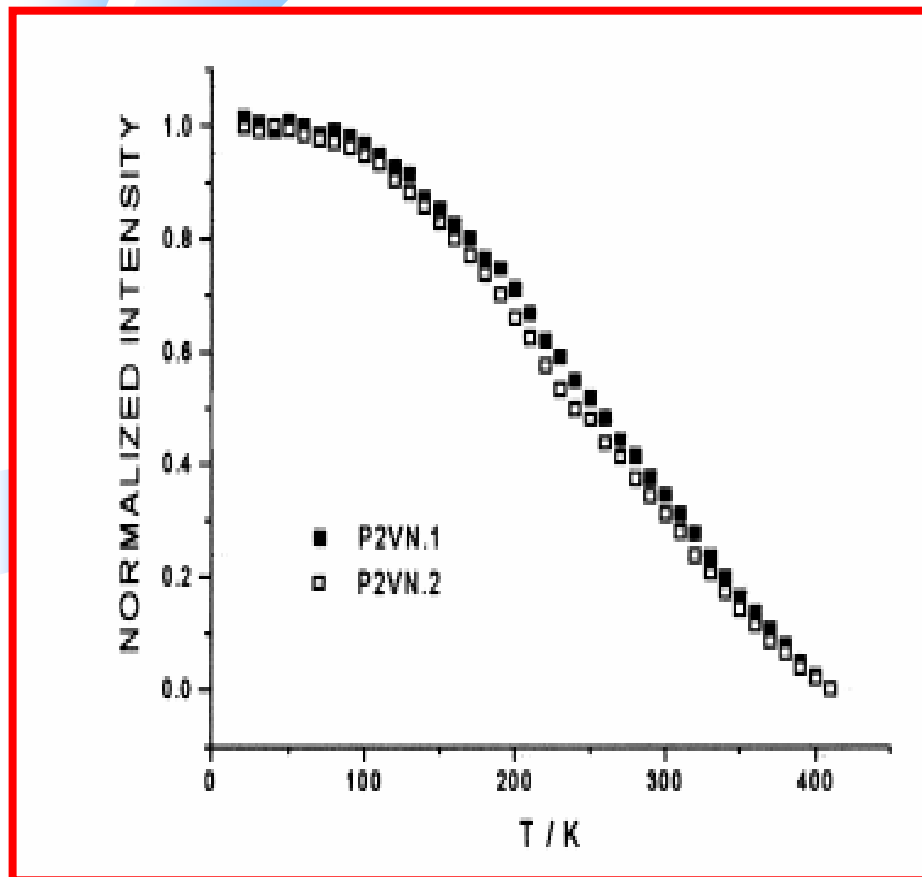


Figure 5. Relative integrated fluorescence ( $\lambda_{ex}$  367 nm) spectral areas  $I_F(T)$  from the NDLDPE/I-I film as a function of temperature during heating from 40 to 400 K. The insert is an expansion of the 200–300 K data.



# Intrinsically luminescent polymers: poly(2-vinyl naphthalene)





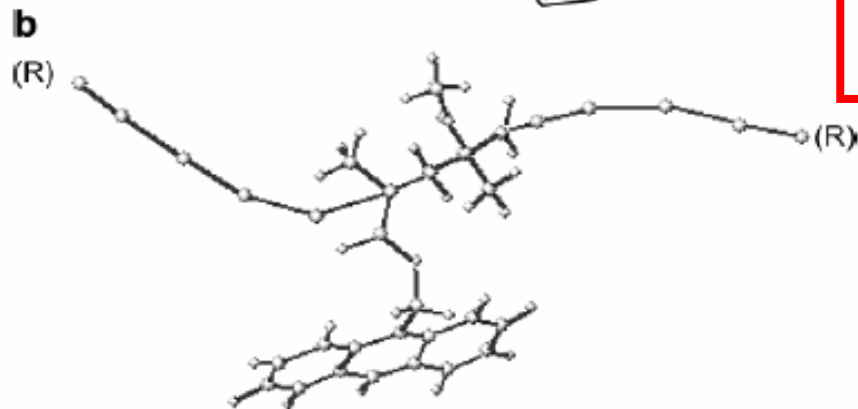
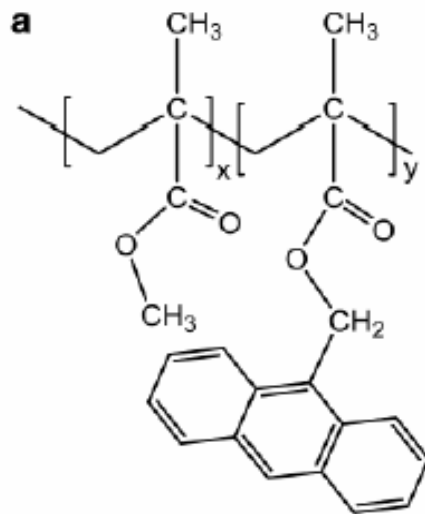
**Figure 9** (a) The temperature dependence and (b,c) Arrhenius plots of the normalized and integrated fluorescence intensity of the P2VN first (-1) and second (-1) heating runs.

**Table II Relaxation Temperatures and Apparent Activation Energies for P2VN by Fluorescence Spectroscopy**

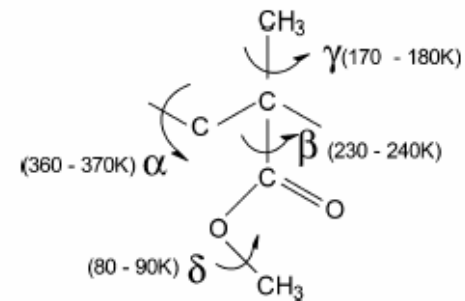
Assignment	P2VN-1		P2VN-2	
	$T$ (K)	$E_a$ (kJ mol <sup>-1</sup> )	$T$ (K)	$E_a$ (kJ mol <sup>-1</sup> )
$\gamma'$ Relaxation	Not observed		110	1.7
$\gamma$ Relaxation	200	4.2	200	3.9
$T_g$	370	67	370	64

$T_g$  (P2VN) = 390 K and  $T_g$  (PB) = 187 K, both determined by DSC; 1 and 2 represent the data for the first and second runs.

# Intrinsically luminescent polymers:



**Figure 1.** (a) Chemical structure of poly(MMA-co-MMAnt).  $x = 98, 35, 25, 10, 4, 2, 0$ ;  $y = 1$ . (b) Optimized geometry of a MMA-MMAnt linkage. The (R) represents the main chain.



**Figure 9.** Illustration of the relaxations detected in PMMA through the technique of fluorescence emission with the variation of the temperature.

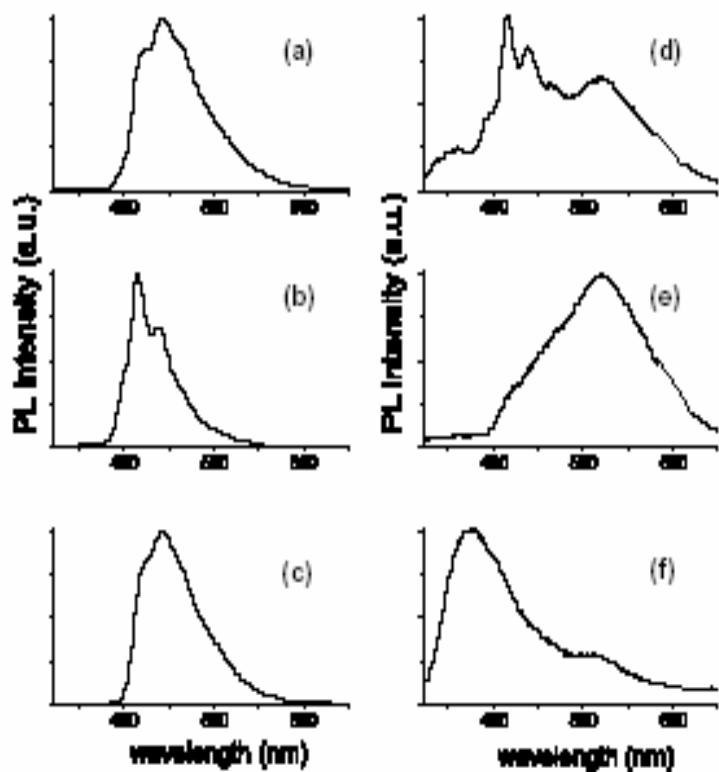


Fig. 8. Solid state fluorescence emission spectra of copolymers poly (MMA-co-MMAnt): (a) 35:1, (b) 25:1, (c) 10:1, (d) 4:1, (e) 2:1 and (f) homopolymer (PMMAnt).  $\lambda_{exc} = 350$  nm.

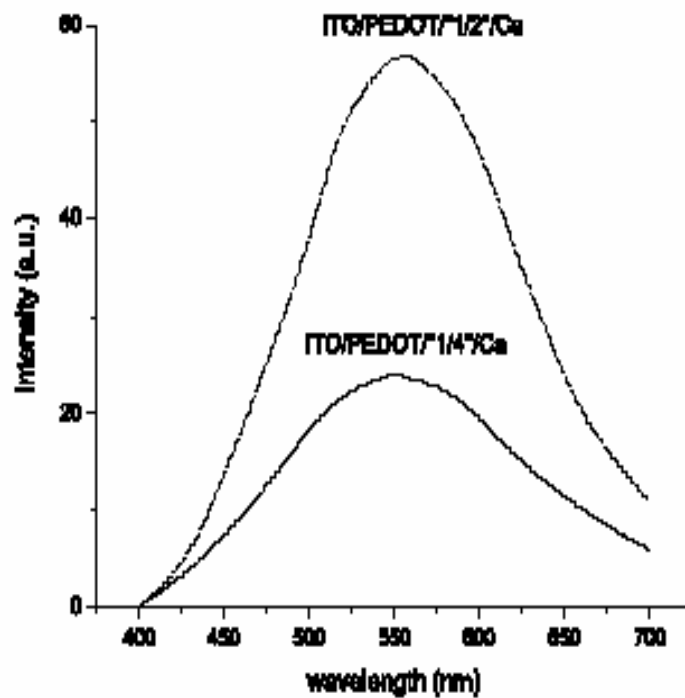
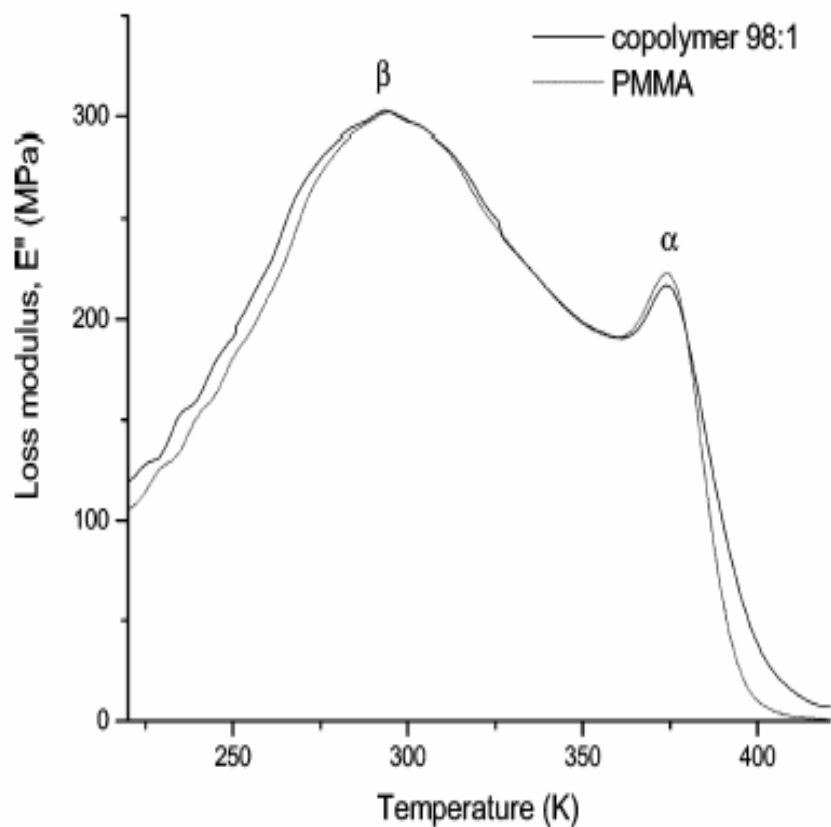
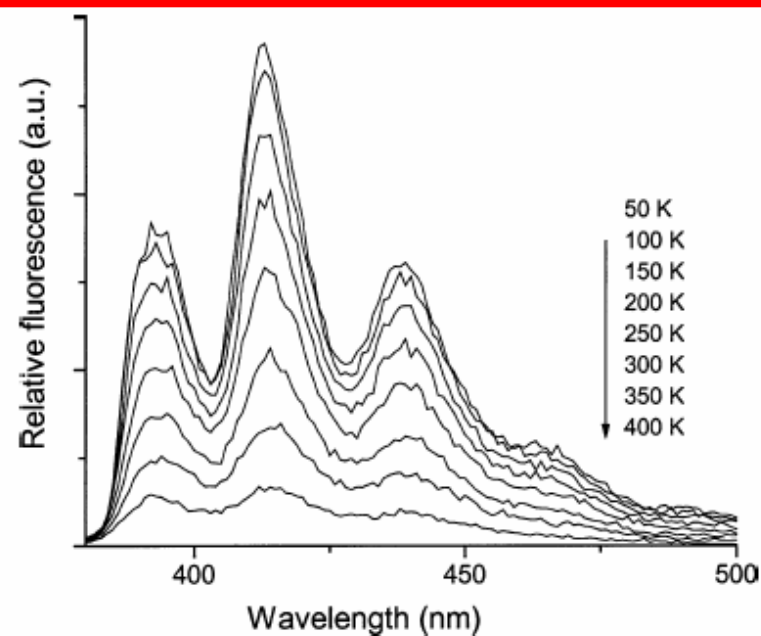


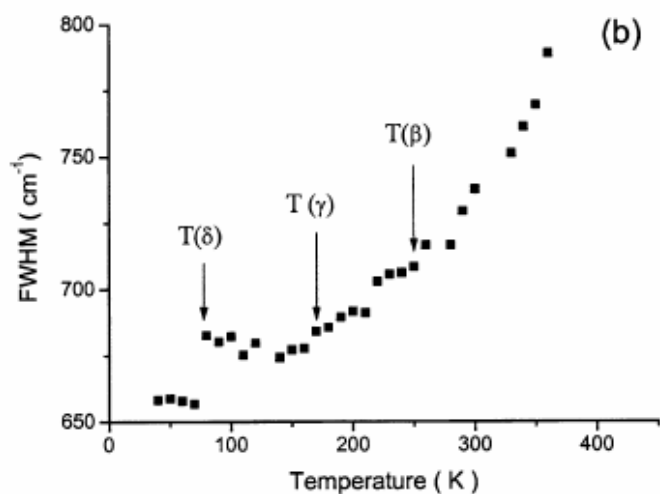
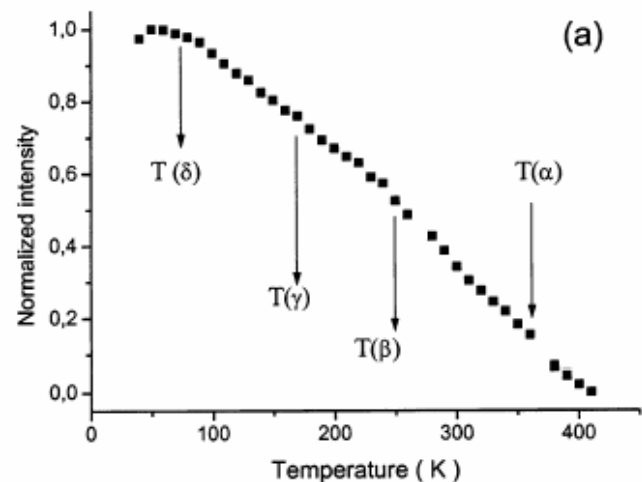
Fig. 10. Electroluminescence of copolymers 2:1 and 4:1. LED configuration: ITO/PEDOT/polymer/Ca/Al.



**Figure 5.** Temperature dependence of loss modulus ( $E''$ ) for poly(methyl methacrylate) and poly(methyl methacrylate-co-9-methylantranyl methacrylate) (98:1). Frequency = 3.3 Hz.

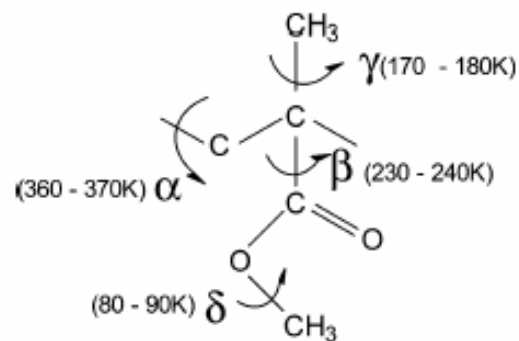


**Figure 6.** Fluorescence spectra of the probe (anthryl group) in poly(methyl methacrylate-co-9-methylantranyl methacrylate) (98:1) at several temperatures.



**Figure 7.** (a) Integrated and normalized fluorescence intensities for poly(methyl methacrylate-co-9-methylantranyl methacrylate) (98:1 molar ratio) vs temperature. (b) Full width at half-maximum (FWHM) of the 0-0 band at the same temperature range.

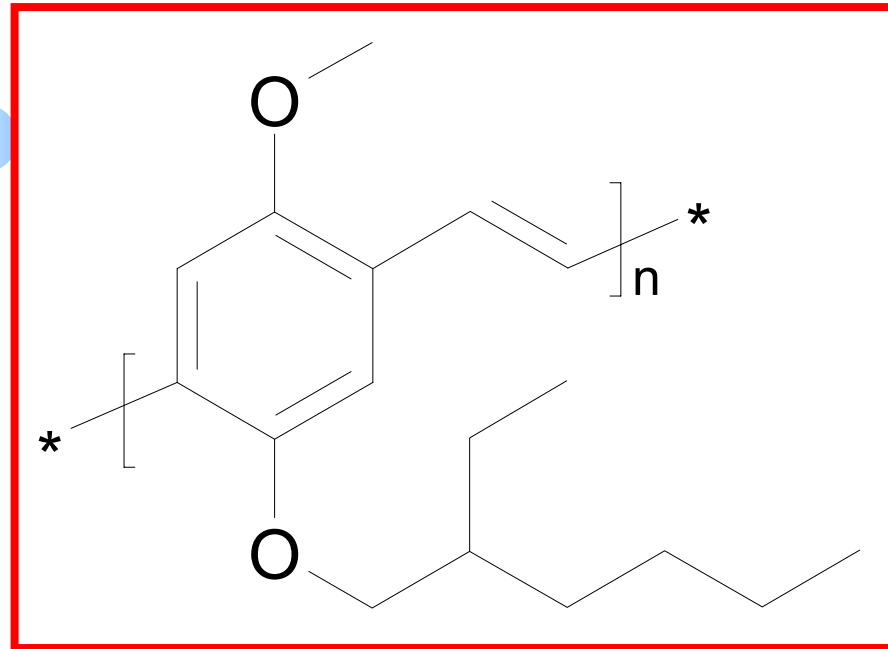
### Relaxations of Poly(methyl methacrylate) E



**Figure 9.** Illustration of the relaxations detected in PMMA through the technique of fluorescence emission with the variation of the temperature.



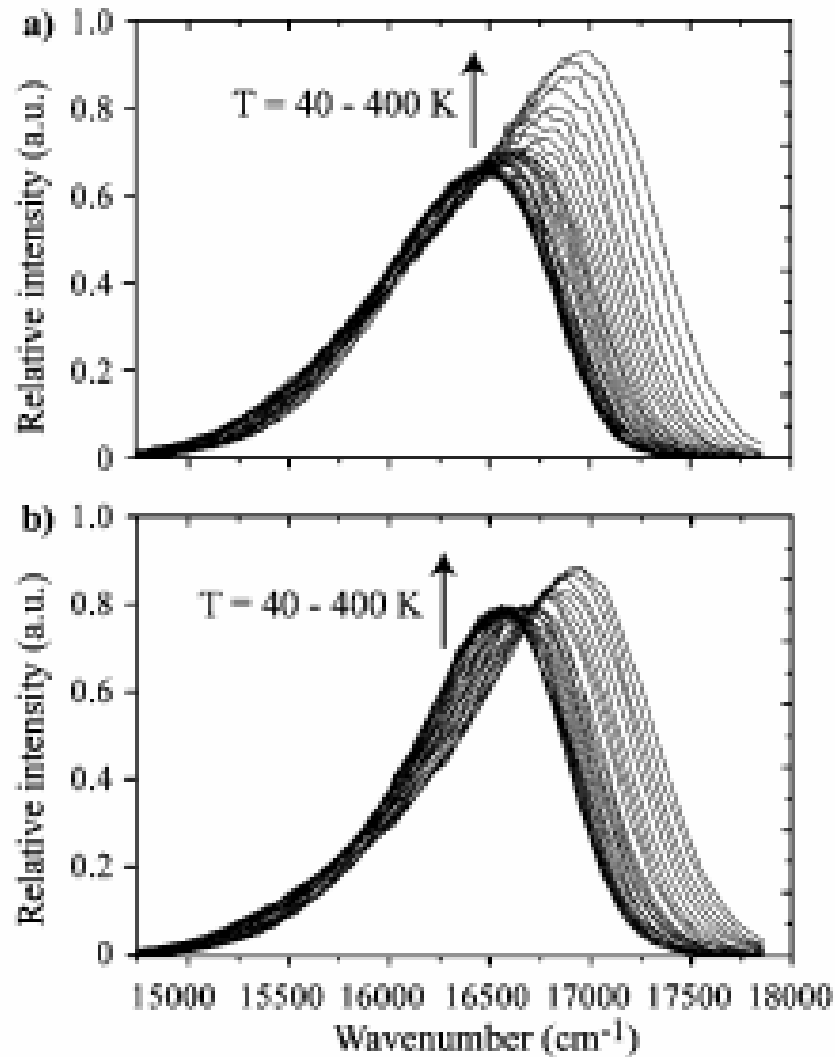
# Conjugated luminescent polymers



Poly(2-methoxy-5-(2-ethylhexyloxy)-p-phenylene vinylene)

MEH-PPV

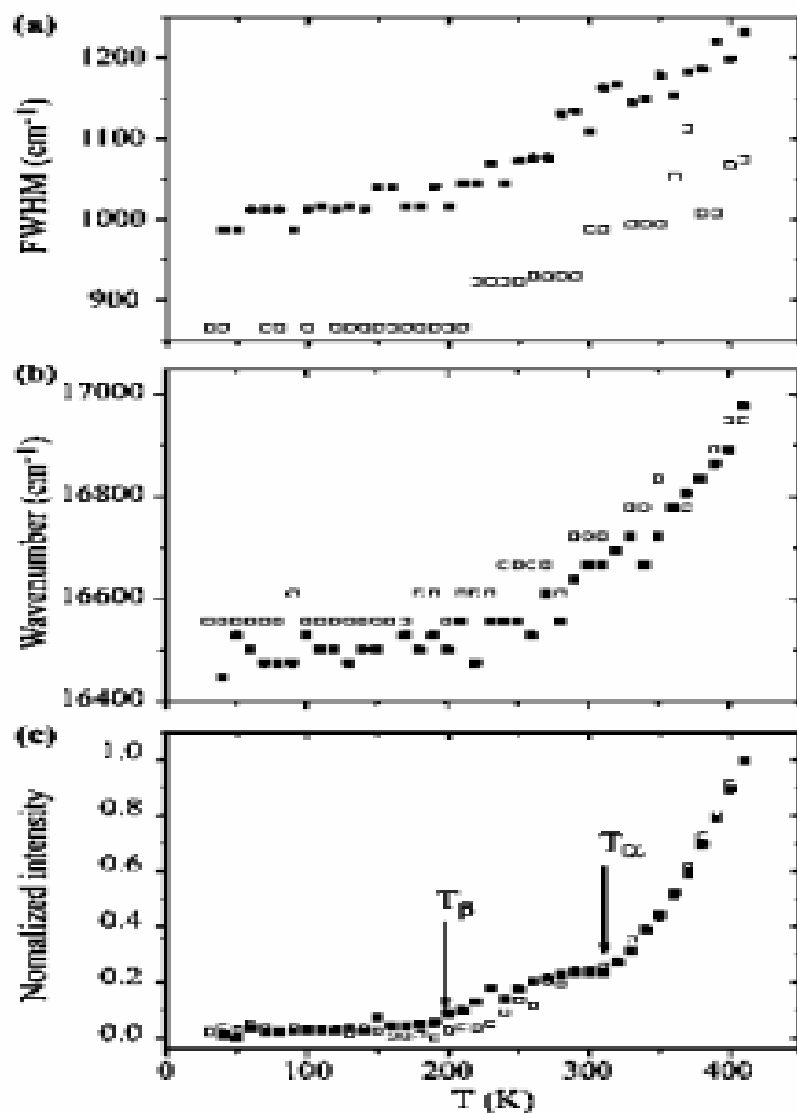
# Steady-state Photoluminescence Emission



**Figure 1.** Photoluminescence spectra at several temperatures (from 40 to 400 K) for MEH-PPV spin-coated films: (a) spun from chloroform; (b) spun from toluene.  $\lambda_{exc} = 21\,270\text{ cm}^{-1}$ .

Blue shift

Increase of intensity



**Figure 2.** (a) Fwhm, (b) photoluminescence peak maxima ( $\text{cm}^{-1}$ ), and (c) integrated and normalized intensity,  $I_P(T)/I_P(T_0)$ , at several temperatures for MEH-PPV films: spun from toluene ( $\square$ ) or spun from chloroform ( $\blacksquare$ ).  $\lambda_{\text{exc}} = 21\,270\text{ cm}^{-1}$ .

# DMTA data

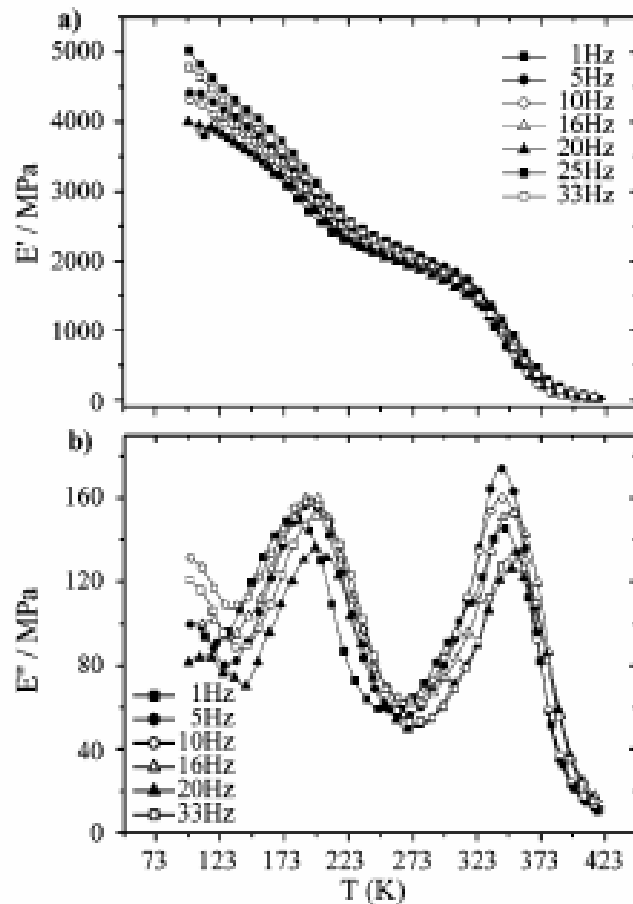


Figure 5. DMTA plots for a film of MEH-PPV at several frequencies.

## Arrhenius plots

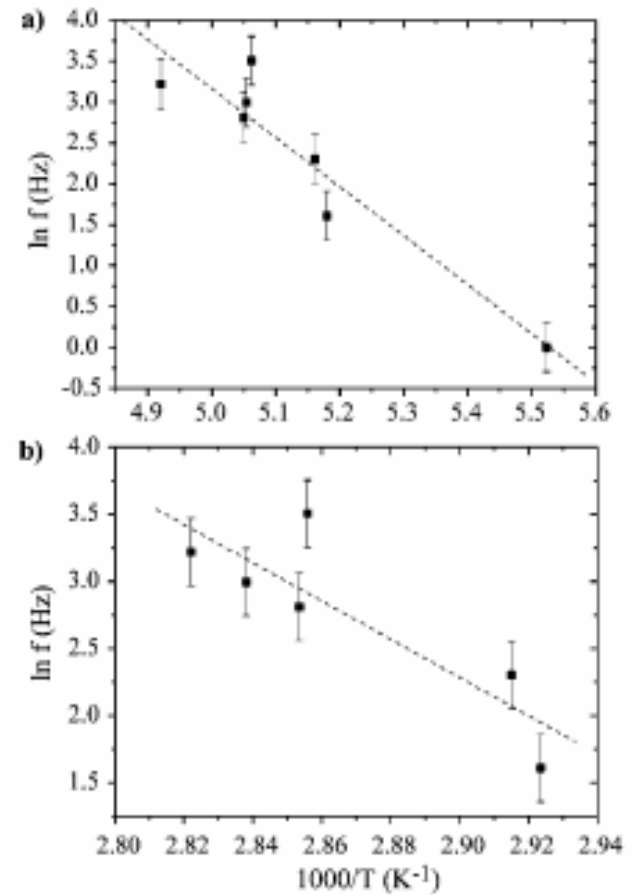
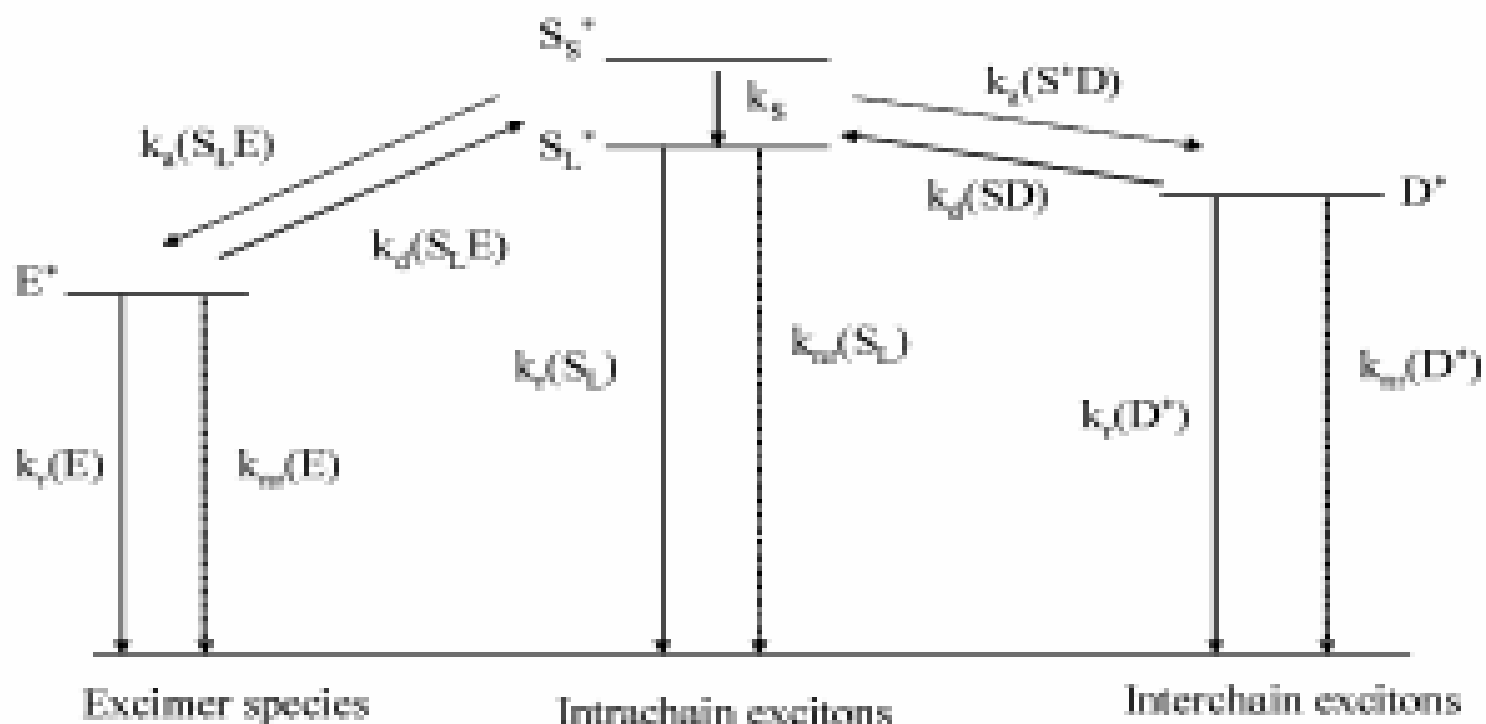


Figure 6. Arrhenius plots for the  $\beta$  (a) and  $\alpha$  (b) relaxations by DMTA.

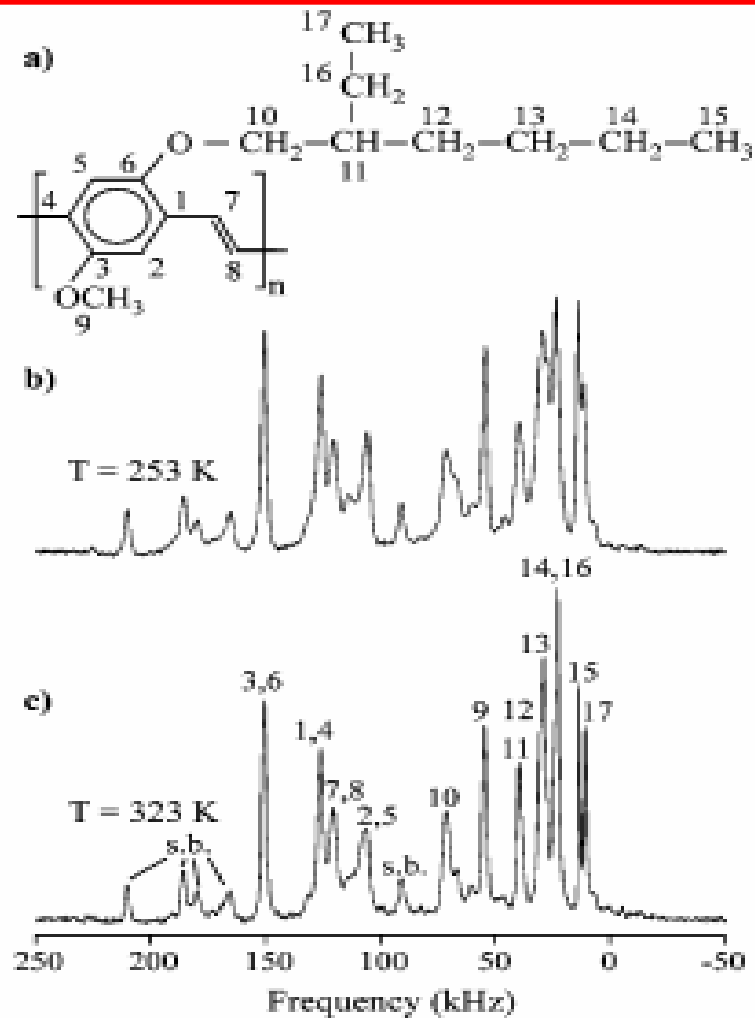


**Figure 3.** Kinetic scheme for the fluorescence decays of MEH-PPV films, where M is the ground-state chromophore,  $M^*$  is an excited-state species, I is the intrachain exciton,  $D^*$  is the excited-state aggregate,  $k_{nr}$  and  $k_r$  are the radiationless and radiative rate constants, respectively,  $k_a(S^*_L D)$  is the rate constant for the intrachain exciton forming an excited dimer, and  $k_d(S^*_L D)$  is the rate constant for the aggregated dissociation re-forming the intrachain exciton.

# Relaxation processes

Techniques	$\beta$ -relaxation	$\alpha$ -relaxation
DMTA	$T = 210 \pm 10 \text{ K}$ $E_a = 29.1 \text{ kJ mol}^{-1} \text{ K}^{-1}$	$T = 330 \pm 10 \text{ K}$ $E_a = 85.4 \text{ kJ mol}^{-1} \text{ K}^{-1}$
TSC	$T = 210 \pm 10 \text{ K}$ $E_a = 22.5 \text{ kJ mol}^{-1} \text{ K}^{-1}$	$T = 310 \pm 10 \text{ K}$ $E_a = 62.4 \text{ kJ mol}^{-1} \text{ K}^{-1}$
Fluorescence	$T = 220 \pm 10 \text{ K}$	$T = 320 \pm 10 \text{ K}$

# Molecular Motions by $^{13}\text{C}$ NMR



**Figure 8.** MEH-PPV repetitive unit (a) and magic-angle-spinning NMR spectra of MEH-PPV with the corresponding line assignments at (b) 253 and (c) 323 K.

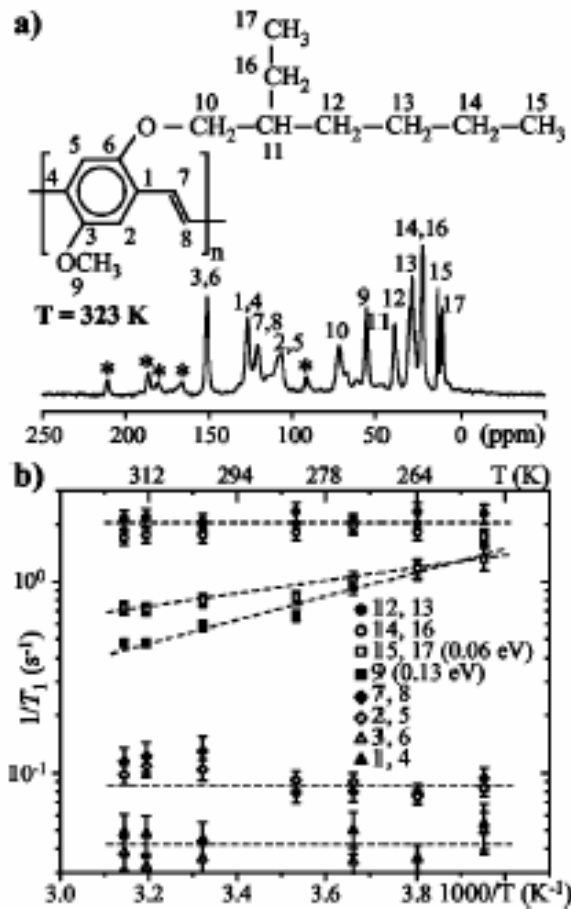


FIG. 1. (a)  $^{13}\text{C}$  CP/MAS NMR spectra of MEH-PPV (\* indicates the spinning side bands). (b) Temperature dependence of the  $^{13}\text{C}$  spin-lattice relaxation rates ( $T_1^{-1}$ ) for all the carbons of MEH-PPV.

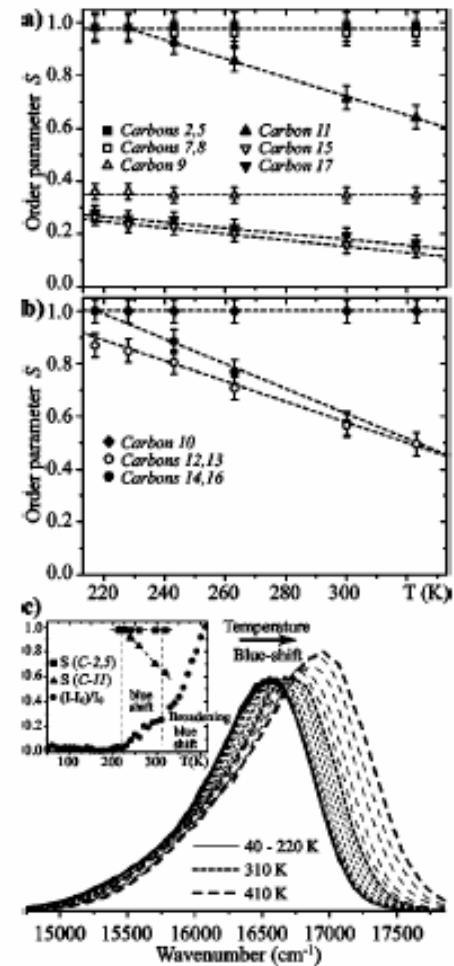
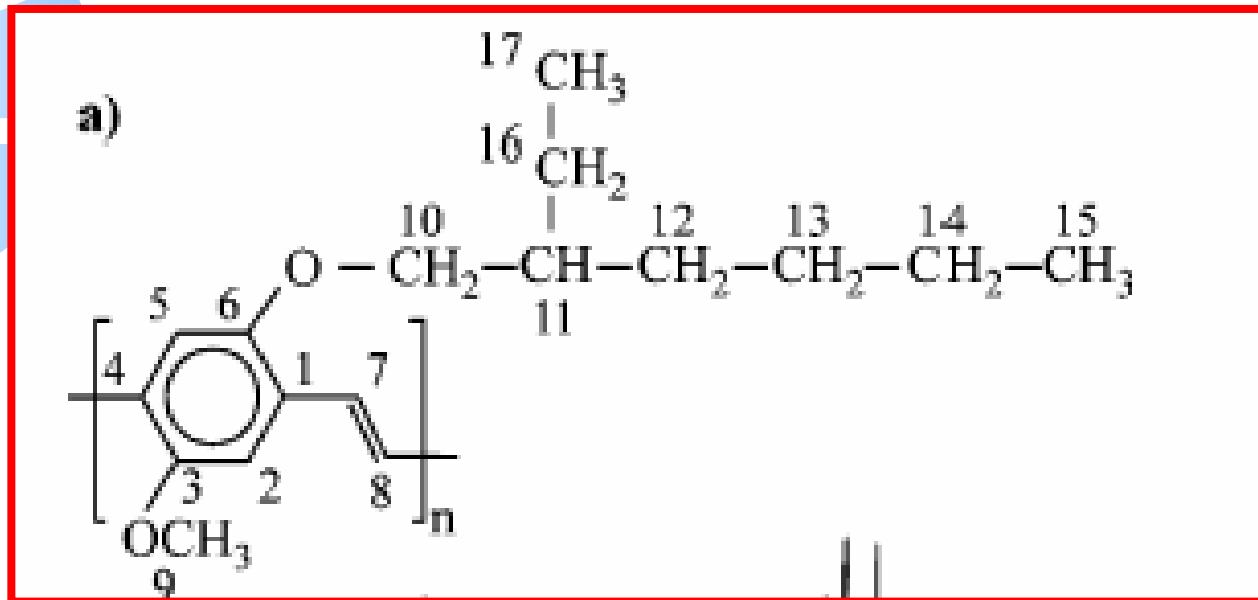


FIG. 2. (a) Plot of the molecular order parameters as a function of temperature for CH and  $\text{CH}_3$  groups in MEH-PPV. (b) Same as in (a) for  $\text{CH}_2$  groups. (c) MEH-PPV PL spectra at several temperatures (from 40 to 410 K). The inset shows the correlation between the differential normalized PL intensities and order parameters at several temperatures.



# Conclusions from $^{13}\text{C}$ RMN



*Carbons 11, 12, 13, 14, 15, 16, and 17 gain mobility after the  $\beta$ -relaxation process*

## Nanorheological approach for characterization of electroluminescent polymer thin films

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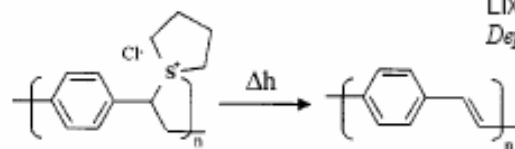
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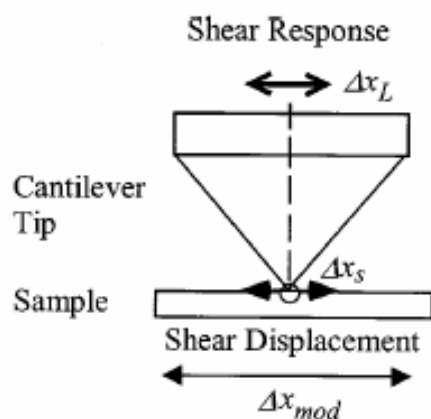
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(a)



(b)

FIG. 1. (a) Thermal conversion of PPV from its sulfonium precursor. (b) Schematic of the SM-SFM. The sample is sinusoidally modulated ( $\Delta x_{mod}$ ) relative to the probing cantilever tip. The shear response ( $\Delta x_L$ ) is a qualitative measure of the contact stiffness (a convoluted expression of the combined shear modulus and the contact area). The contact stiffness is represented by the sample deformation  $\Delta x_S$ , and is directly obtained by the cantilever response  $\Delta x_L$ , with  $\Delta x_{mod} = \Delta x_L + \Delta x_S$ .

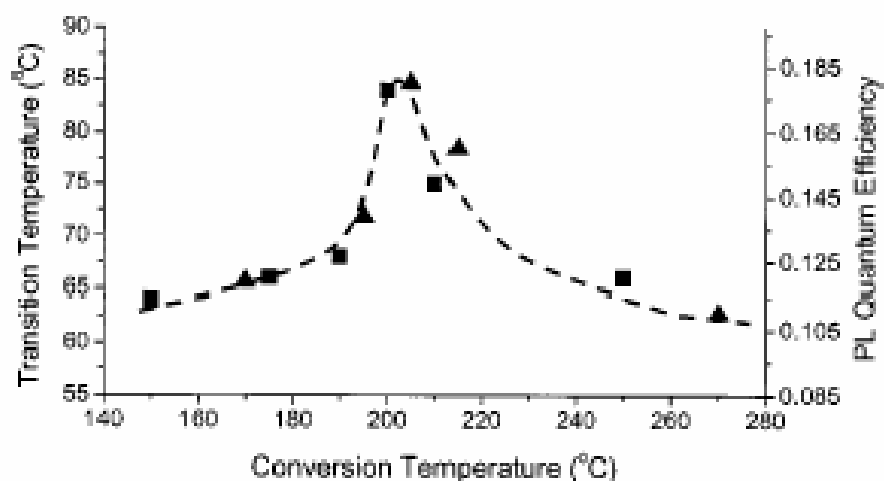


FIG. 3. Qualitative comparison of the PL efficiency ( $\blacktriangle$ ) (see Ref. 7) with the rheological transition ( $\blacksquare$ ) as function of the PPV conversion temperature. The rheological transitions were measured with the SM-SFM method. A dashed line has been added for trend enhancement.

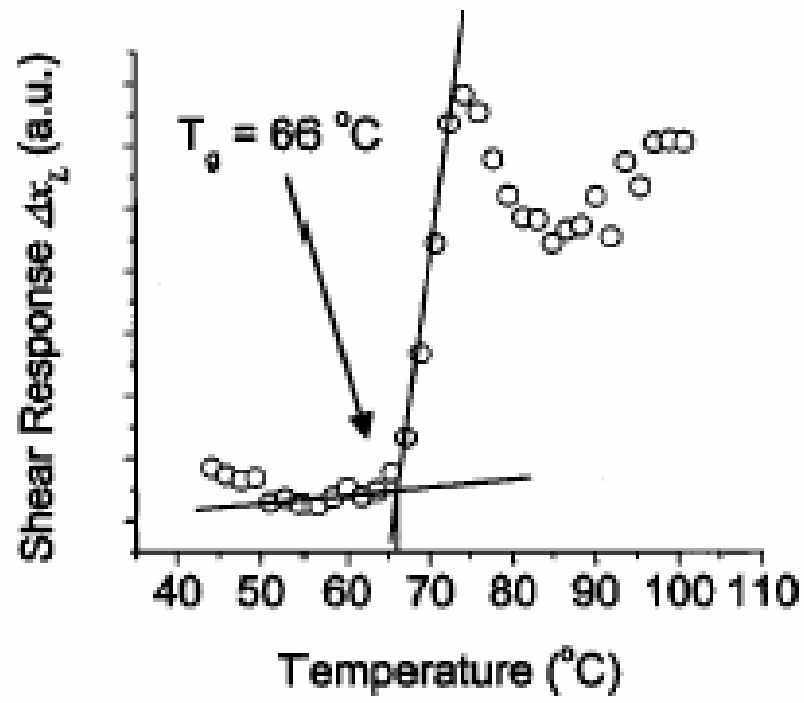


FIG. 2. Representative thermal-rheological SM-SFM plot of PPV at a conversion temperature  $T_{conv}$  of 175  $^{\circ}\text{C}$ . At  $T_g$ , the contact area increase dominates the modulus decrease of the polymer. Thus, the contact stiffness increases above  $T_g$  until the transition process is complete.



# Implications of the relaxation processes on the electroluminescence properties

# Temperature-dependent device model for polymer light-emitting diodes: significance of barrier height

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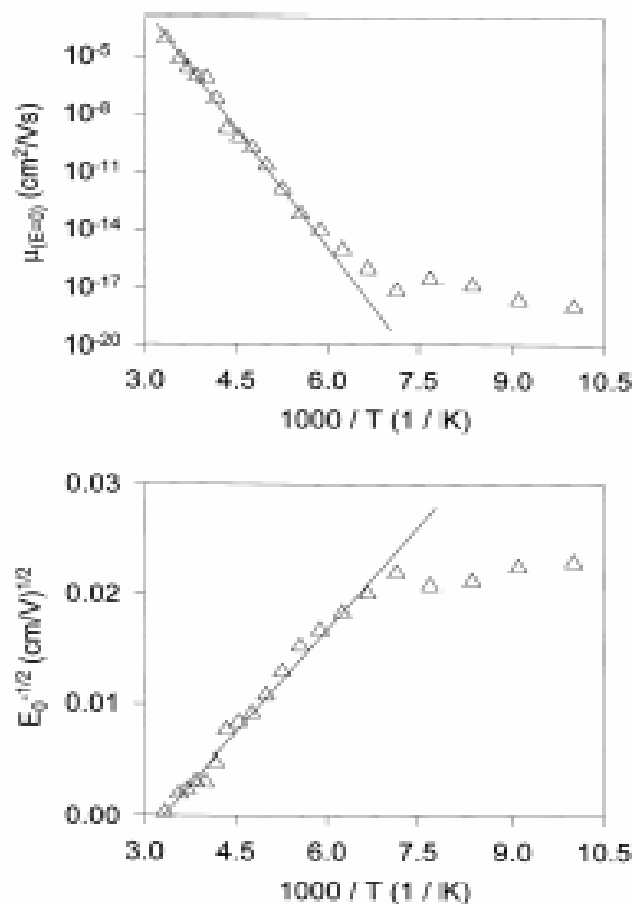


Fig. 2. Variation of the mobility parameters with temperature.

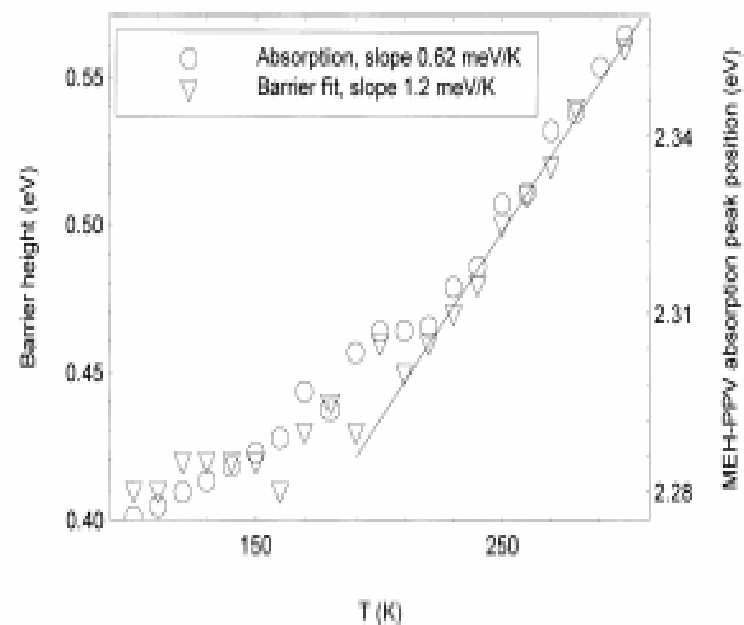
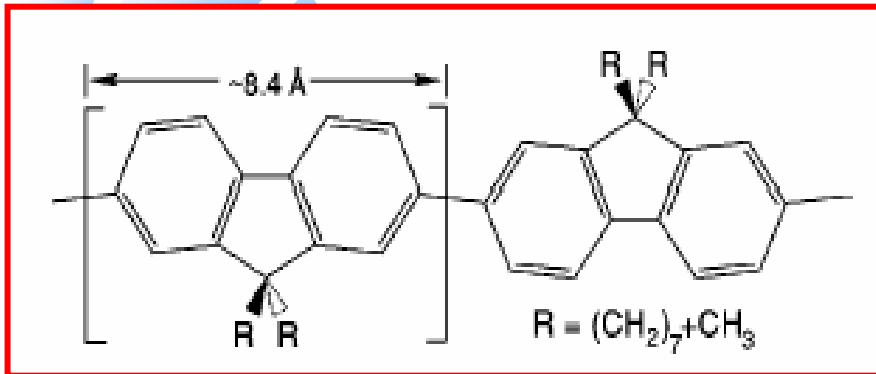


Fig. 3. Decrease of barrier height for hole injection with temperature compared to the red-shift of the absorption of MEH-PPV.

# Electroluminescence and photoluminescence



poly(9,9-(di-*n,n*-octyl-fluorene))

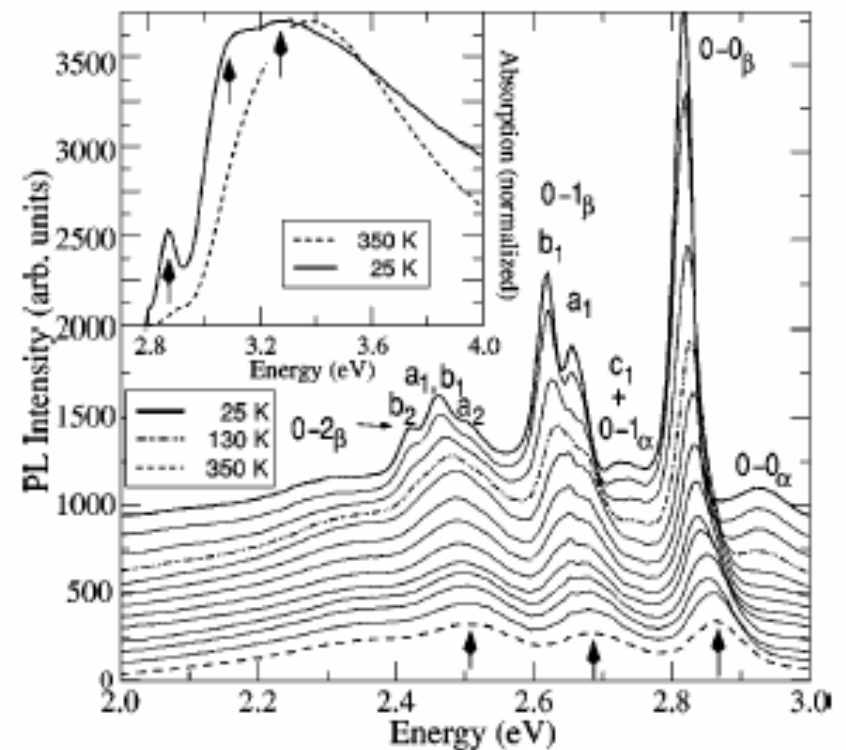
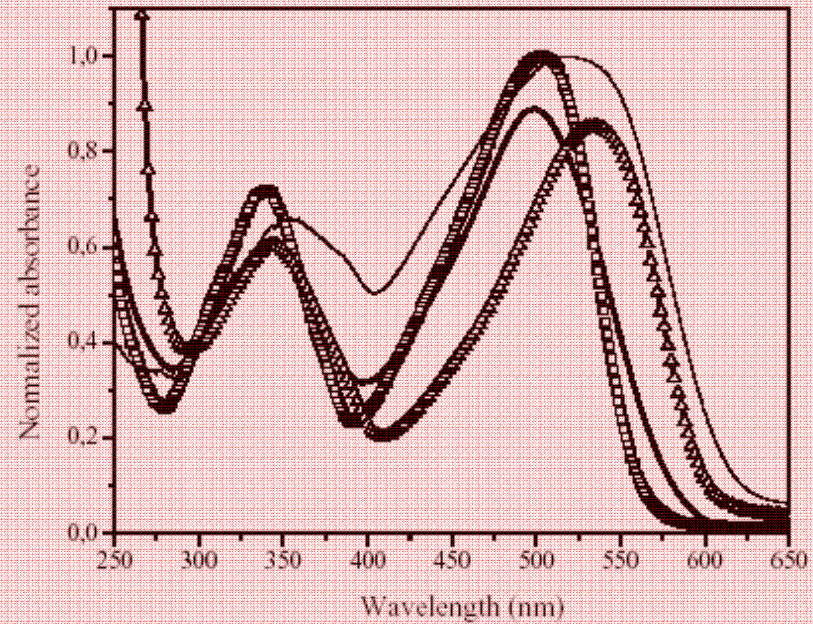
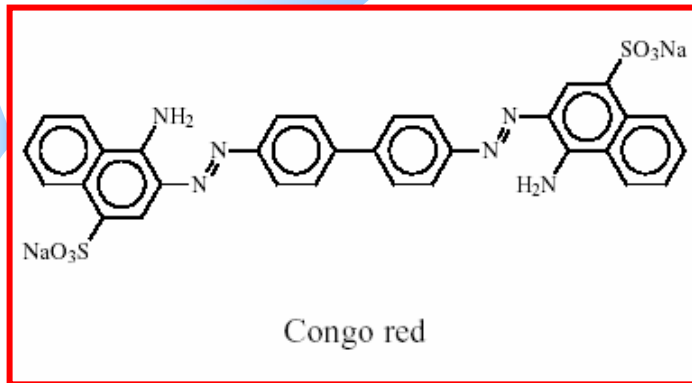


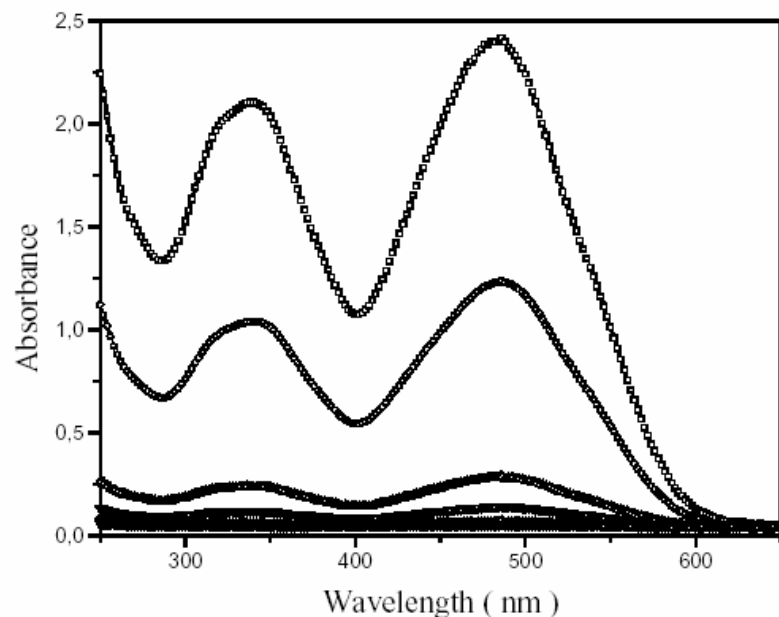
FIG. 2. Progression of PF8 sample *A* photoluminescence spectra on cooling from 350 K ( $\lambda_{\text{Ex}} = 390$  nm). All PL curves are self-absorption corrected and offset for clarity. Inset: Two corresponding absorption spectra with arrows indicating the  $\beta$ -phase 0-0, 0-1, and 0-2 vibronic bands.

# Spectral profiles and condensed medium

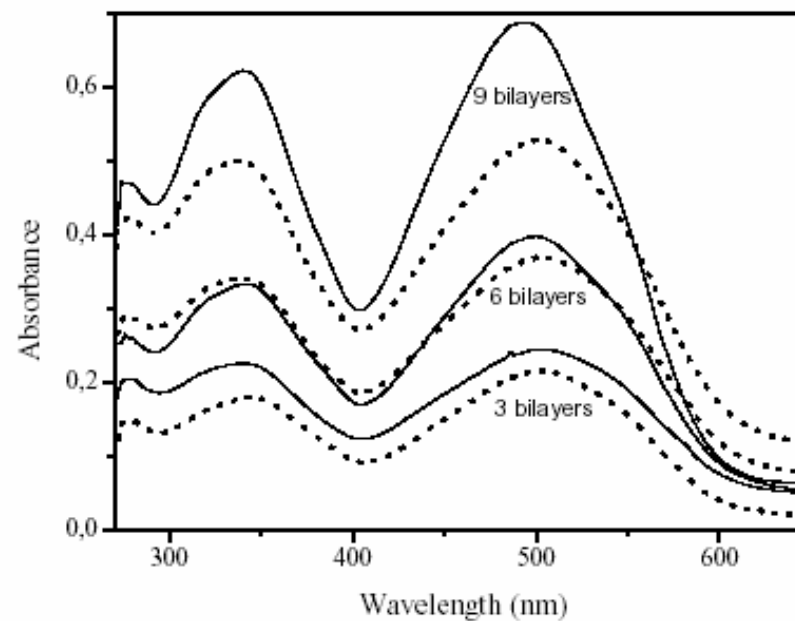
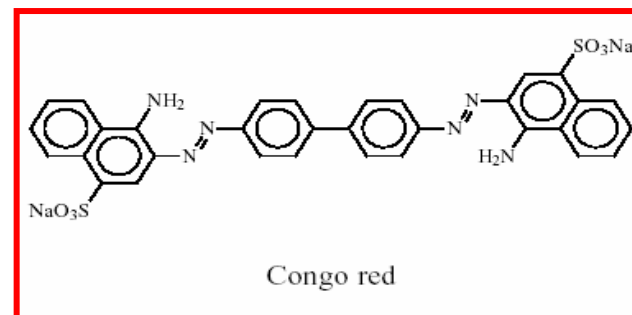


**Figure 2.** Electronic absorption spectra of 10<sup>-3</sup> mol L<sup>-1</sup> solutions (by transmittance) of CR (●) in deionized water, (□) MeOH, (△) DMSO and (○) CR adsorbed on cellulose fibers (by diffuse reflectance).

# Spectral profiles and condensed medium



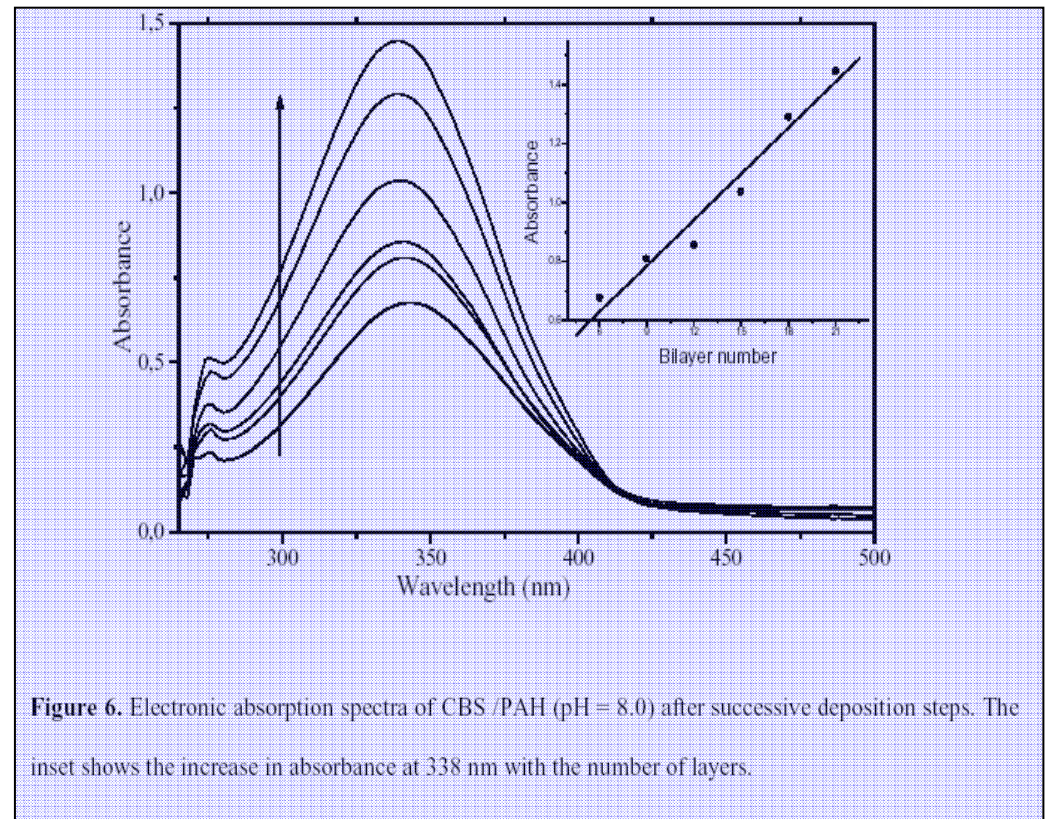
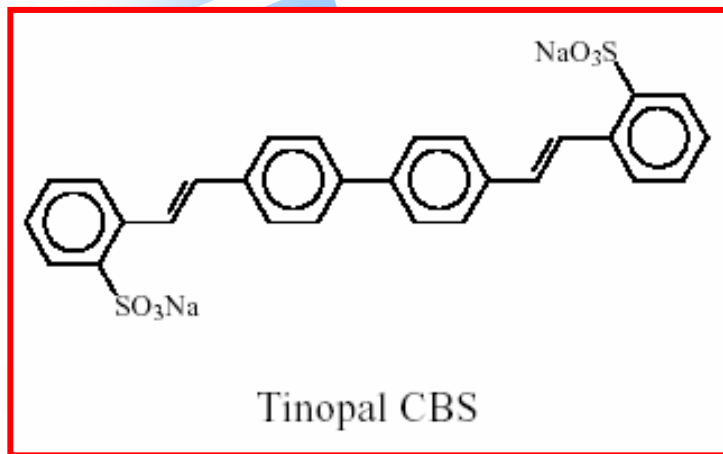
**Figure 3. a.** Electronic absorption spectra CR in aqueous solution of CR: (—\*—\*)  $7.2 \cdot 10^{-6}$  mol L<sup>-1</sup>, (—◇—◇—)  $1.4 \cdot 10^{-5}$  mol L<sup>-1</sup>, (—▽—▽—)  $7.2 \cdot 10^{-5}$  mol L<sup>-1</sup>, (—△—△—)  $1.4 \cdot 10^{-4}$  mol L<sup>-1</sup>, (—○—○—)  $7.2 \cdot 10^{-4}$  mol L<sup>-1</sup> and (—□—□—)  $1.4 \cdot 10^{-3}$  mol L<sup>-1</sup>.



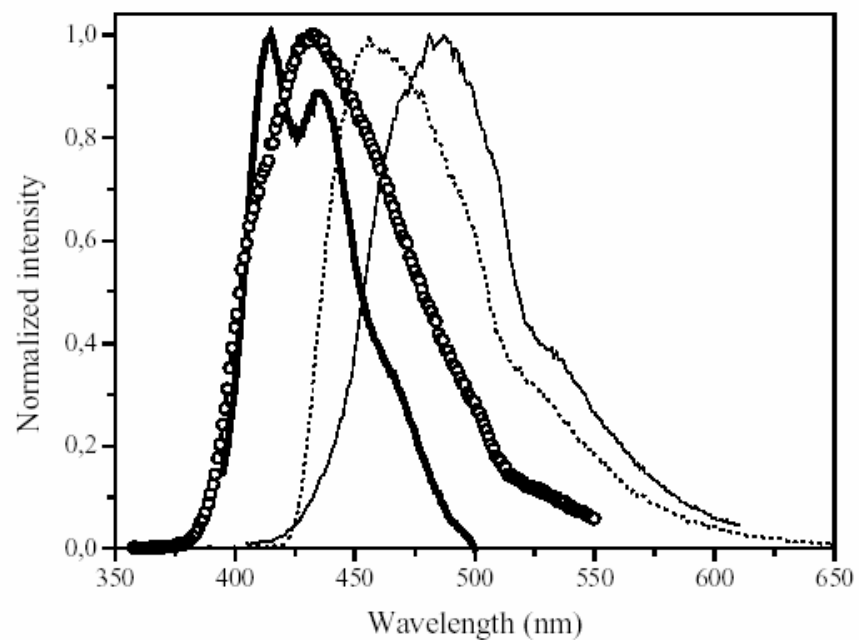
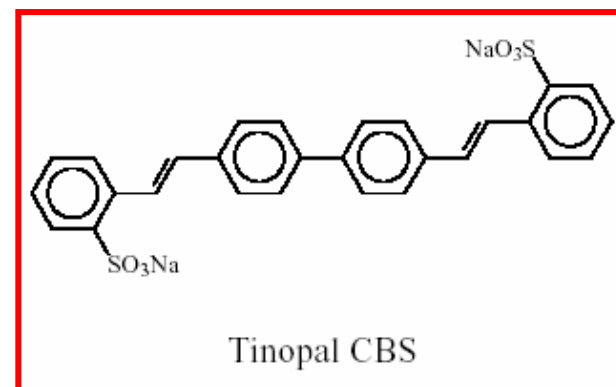
**Figure 4.** Electronic absorption spectra of CR/PAH LbL films prepared using dipping solution with (---) pH 2.0 and (—) pH 10.0.



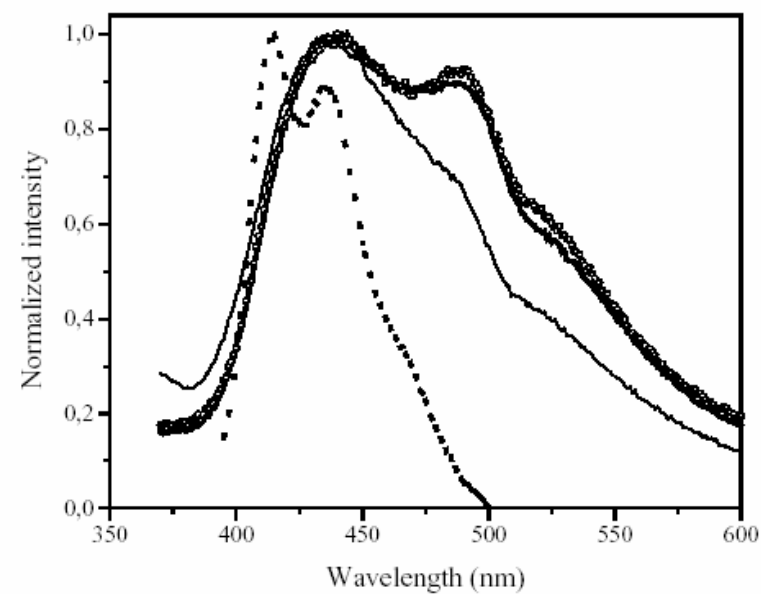
## Spectral profiles and condensed medium



**Figure 6.** Electronic absorption spectra of CBS /PAH (pH = 8.0) after successive deposition steps. The inset shows the increase in absorbance at 338 nm with the number of layers.



**Figure 7.** Fluorescence spectra of CBS in: aqueous solutions ( $10^{-7}$  mol L $^{-1}$ ) (-o-o-o) and ( $10^{-2}$  mol L $^{-1}$ ) (—), in a crystal (---) and adsorbed on cellulose fibers (—).  $\lambda_{exc}$  = 348 nm.



**Figure 8.** Fluorescence spectra of CBS adsorbed on cellulose fibers (----) and on LbL films prepared using dipping solution with pH 4.0 (-o-o-o) and pH 10.0 right after deposition (—) and after 20 minutes (—).

# Relaxation process and spectral broadening

**Table 1.** Some Properties of Native Polyethylene Films Employed

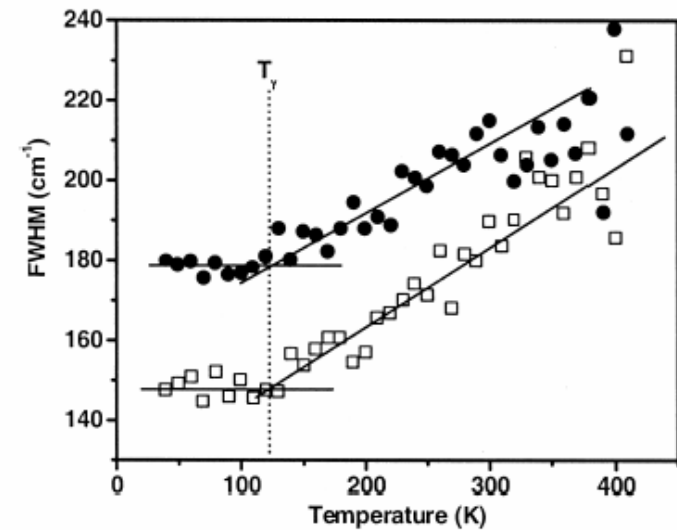
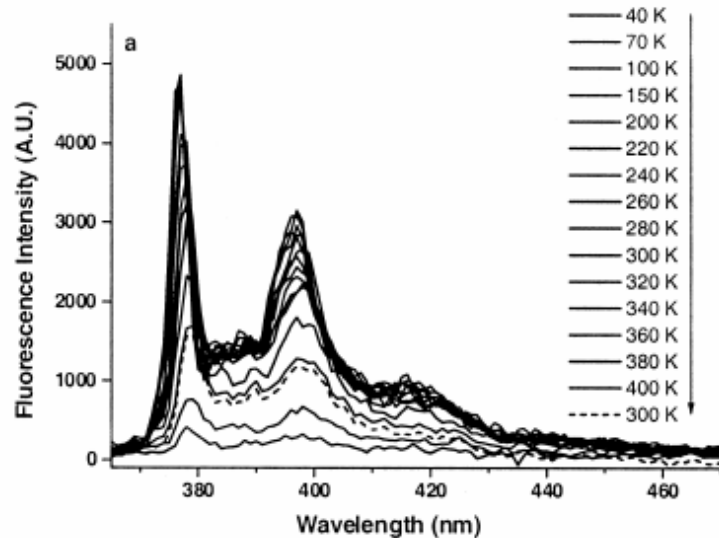
Symbol	Thickness ( $\mu\text{m}$ ) <sup>32</sup>	$T_m$ (K) <sup>a</sup>	$T_g$ (K) <sup>a</sup>	Density ( $\text{g}/\text{cm}^3$ )	% Crystallinity <sup>b</sup>
PE32	32	368	—	0.900 <sup>32</sup>	32 (12)
PE42	38	389, 394 <sup>c</sup>	243	0.918	42 (42)
PE42 <sup>d</sup>	42	383	—	—	42 (41)
PE76	13	403	—	0.952	76 (42)

<sup>a</sup> Heat flow maxima from endotherms of DSC (heating) thermograms.

<sup>b</sup> By X-ray diffraction. Values in parentheses from DSC analyses.

<sup>c</sup> A polymer blend; both heat flow maxima in DSC thermograms are reported.

<sup>d</sup> A native PE42 film bombarded with 7.0 MeV alpha particles at a dose of 120 kGy.



**Figure 7** FWHM of the 1-pyrenyl 0–0 emission band from  $10^{-2}$  mol/kg pyrene in **PE32** (●) and in **PE76** (□) films that were bombarded with 7.0 MeV alpha particles at doses = 110 (**PE32**) and 87 kGy (**PE76**). The vertical dotted line indicates the onset temperature for the  $\gamma$ -relaxation, based on the intersection of the best-fit straight line segments. The data for the **Py-PE32** sample have been offset by  $30 \text{ cm}^{-1}$  for clarity.

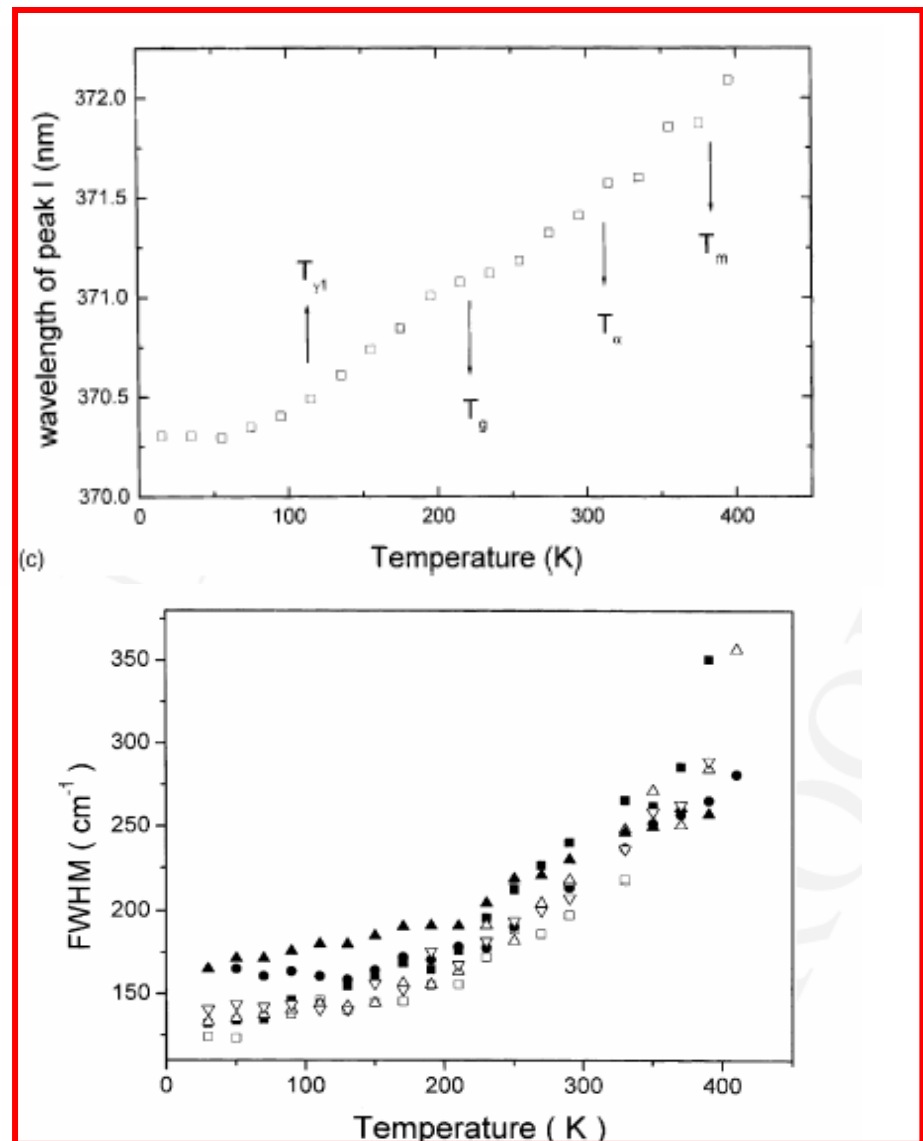
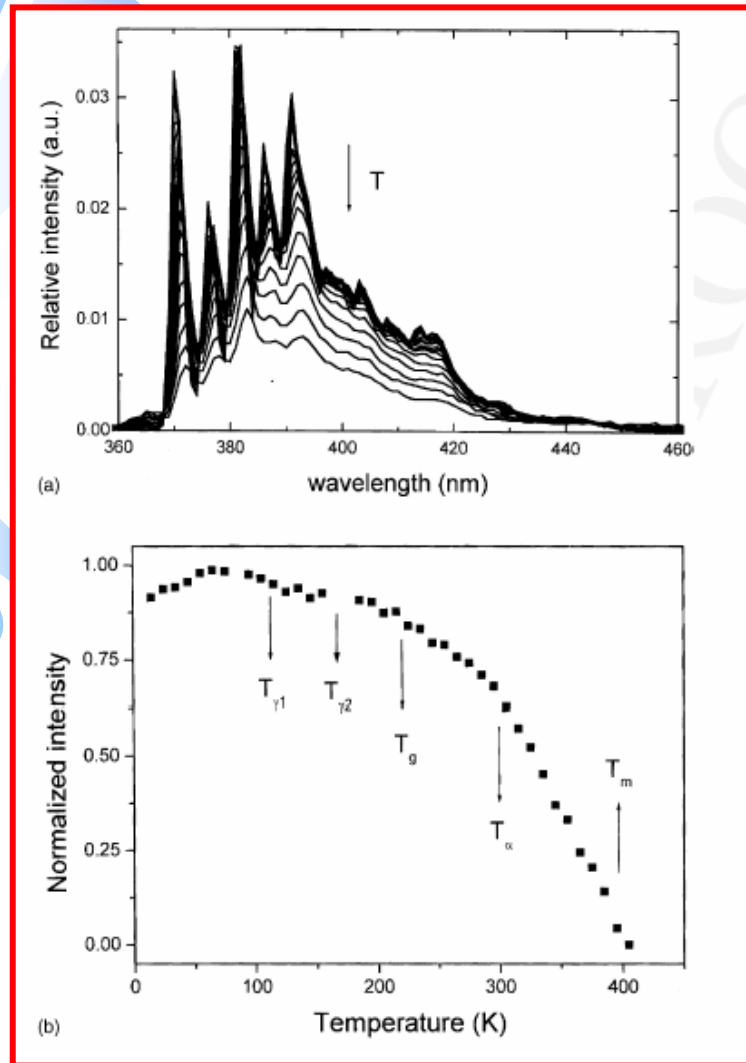


Fig. 1. (a) Fluorescence spectra of pyrene sorbed in HDPE; (b) integrated intensity of the fluorescence band; (c) wavelength of the vibrational band I ( $\lambda_{em} = 372\text{--}374\text{ nm}$ ). Several temperatures between 15 and 405 K.

Martins, et. Al. J. Photochem. Photobiol. A Chem. 2002

# Thank you!

Fred

Miguel

Jennifer

Summer School - Stereochemical Aspects of Novel Materials, UCSB, august 14-27, 2005